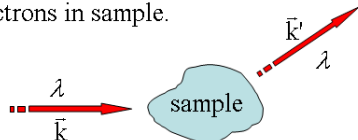


Appendix : X-ray diffraction

(I) General

X-ray diffraction process involves only **elastic scattering** of x-ray photons with electrons in sample.

$$k = k' = \frac{2\pi}{\lambda}$$



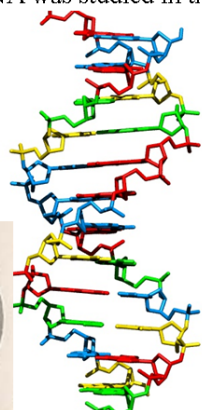
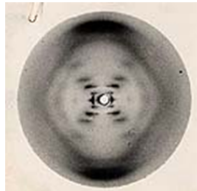
- It is the most important tool to characterize the crystal structure of a sample, including
 - type of crystal structure, location of atoms in unit cell, chemical phase
 - microstructure (single crystal/polycrystalline/amorphous, grain size,...)
 - strain,...

- Complicated crystal structure can be determined by x-ray diffraction.

e.g. The double helix structure of DNA was studied in the beginning of the 1950s.

See "The Discovery of the Molecular Structure of DNA – The Double Helix" at

http://nobelprize.org/medicine/educational/dna_double_helix/readmore.html



(II) Principle

(A) X-ray photon scattered by an electron (see Cullity, ch.3)
elastic scattering (first studied by J.J. Thomson)
(Note: Compton scattering is inelastic.)

(B) X-ray photon scattered by an atom

- not scattered by nucleus because of its heavy mass
- scattered elastically & coherently by all electrons

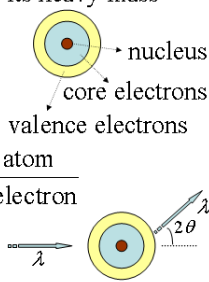
The scattering efficiency of an atom is measured by the **atomic scattering factor**:

$$f = \frac{\text{amplitude of wave scattered by an atom}}{\text{amplitude of wave scattered by one electron}}$$

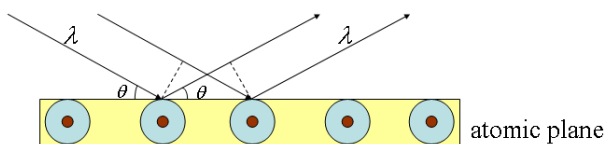
f depends on $\sin \theta / \lambda$

(f of different elements is tabulated:

<http://www.structure.llnl.gov/xray/comp/scatfac.htm>)



(C) X-ray photon scattered by a plane of atoms in a lattice

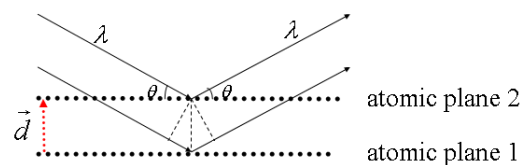


Just like a mirror, incident angle = reflection angle = θ
There is no path difference in the two rays. So the two rays have the same phase. The reflection (or **diffraction**) intensity depends on the distribution of atoms in the plane and the atomic scattering factor(s) of the atom(s) and the angle θ .
The diffraction amplitude for an atomic plane

$$A \propto \sum_{i\text{-th atom in one unit cell in the plane}} f_i \equiv F$$

planar structure factor

(D) X-ray photon scattered by parallel planes of a crystal



The two rays have a path difference = $2d \sin \theta$

Hence the phase difference ϕ : $\frac{\phi}{2\pi} = \frac{2d \sin \theta}{\lambda}$

or $\phi = 2k d \sin \theta = \Delta k \cdot \vec{d}$

$$\vec{k} = \Delta \vec{k} + \vec{k}'$$

$$(k = k' = \frac{2\pi}{\lambda})$$

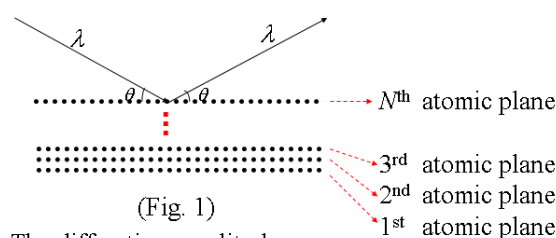
$$\Delta k = 2k \sin \theta$$

Note: $\Delta \vec{k}$ is normal to the planes.

Then the diffraction amplitude due to these two planes

$$\propto F_1 + F_2 e^{i\Delta k d}$$

For a crystal with N atomic planes:



The diffraction amplitude

$$A \propto F_1 + F_2 e^{i\Delta k d} + F_3 e^{i\Delta k 2d} + \dots + F_N e^{i\Delta k (N-1)d}$$

(Obviously, in general $A \propto \sum_{\text{all atoms in sample}} f_i e^{i\Delta \vec{k} \cdot \vec{r}_i}$)

Assume all planes are identical with planar structure factors:

$$F_1 = F_2 = \dots = F_N \equiv F$$

Then

$$A \propto F \sum_{p=0}^{N-1} e^{ip\Delta k d} = F \frac{1 - e^{iN\Delta k d}}{1 - e^{i\Delta k d}} = F \frac{\sin(N\Delta k d / 2)}{\sin(\Delta k d / 2)} e^{i(N-1)\Delta k d / 2}$$

The diffraction intensity

$$I \propto |A|^2$$

$$\propto |F|^2 \frac{\sin^2(N\Delta k d / 2)}{\sin^2(\Delta k d / 2)}$$

which is maximum ($= |F|^2 N^2$) if $\frac{\Delta k d}{2} = n\pi$

or $2d \sin \theta = n\lambda$ $n = 1, 2, \dots$ **Bragg equation** of diffraction

This is the condition for observing a diffracted beam.

A sample (single phase) is characterized by

- crystal type
 - lattice parameters: a b c and angles α β γ
- Crystal planes can be labeled by (hkl) .

Bragg equation for diffraction due to (hkl) crystal planes:

$$2d_{hkl} \sin \theta = \lambda$$

where d_{hkl} is the interplanar spacing ($\equiv$ the perpendicular separation between parallel (hkl) crystal planes).

d_{hkl} can be calculated with formulae tabulated in reference book. (e.g. B.D. Cullity, *Elements of x-ray diffraction*, 2/e, QC482.D5C84 1978).

The Bragg law can be rewritten as

$$\Delta k = \frac{2\pi}{d_{hkl}}$$

In general

$$\Delta \vec{k} = \vec{G}_{hkl}$$

Here we use the reciprocal lattice vector $\vec{G}_{hkl} = h\vec{a}^* + k\vec{b}^* + \ell\vec{c}^*$

For a crystal lattice, the reciprocal lattice is constructed with fundamental vectors defined as:

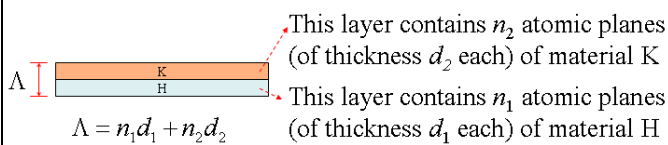
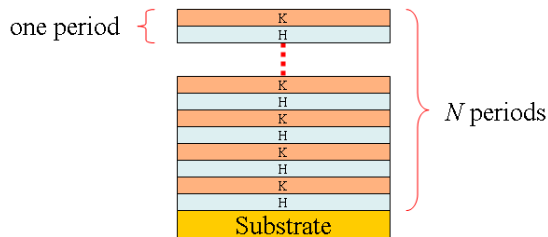
$$\vec{a}^* = \frac{2\pi\vec{b} \times \vec{c}}{\vec{a} \cdot \vec{b} \times \vec{c}} \quad \vec{b}^* = \frac{2\pi\vec{c} \times \vec{a}}{\vec{a} \cdot \vec{b} \times \vec{c}} \quad \vec{c}^* = \frac{2\pi\vec{a} \times \vec{b}}{\vec{a} \cdot \vec{b} \times \vec{c}}$$

To understand this principle, consider reciprocal lattice of cubic structure or 2D crystals*.

(* B.N. Narahari Achar, Am. J. Phys. **54**, 663 (1986))

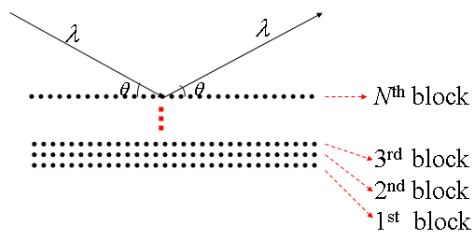
(E) X-ray diffraction of a superlattice

Structure of a superlattice composed of two materials H & K:



We also call this sample a **composition-modulated structure** because its composition is “modulated” (i.e., changed periodically) in one dimension (usually normal to the substrate surface) and the superlattice period or modulation wavelength: $\Lambda = n_1 d_1 + n_2 d_2$

If we consider each period as a block and replace the N atomic planes in Fig. 1 by the N blocks, then



the diffraction amplitude

$$A \propto F'_1 + F'_2 e^{i\Delta k \Lambda} + F'_3 e^{i2\Delta k \Lambda} + \dots + F'_N e^{i(N-1)\Delta k \Lambda}$$

All blocks are identical with block structure factors

$$F'_1 = F'_2 = \dots = F'_N \equiv F'$$

$$\text{Then } A \propto F' \sum_{p=0}^{N-1} e^{ip\Delta k \Lambda} = F' \frac{1 - e^{iN\Delta k \Lambda}}{1 - e^{i\Delta k \Lambda}} = F' \frac{\sin(N\Delta k \Lambda / 2)}{\sin(\Delta k \Lambda / 2)} e^{i(N-1)\Delta k \Lambda / 2}$$

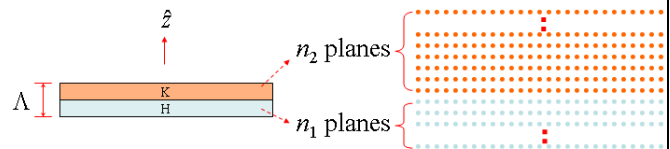
The diffraction intensity

$$I \propto |A|^2 \propto |F'|^2 \frac{\sin^2(N\Delta k \Lambda / 2)}{\sin^2(\Delta k \Lambda / 2)}$$

which is maximum ($=|F'|^2 N^2$) if $\frac{\Delta k \Lambda}{2} = n\pi$

or $2\Lambda \sin \theta = n\lambda \quad n = 1, 2, \dots$

This diffraction condition can tell how we find the diffraction beams but their intensities are also determined by $|F'|^2$.



The first atomic plane of H is at $z = 0$ (assumed).

The first atomic plane of K is at $(n_1 - 1)d_1 + \frac{1}{2}(d_1 + d_2)$

$$= n_1 d_1 + \frac{1}{2} \Delta d$$

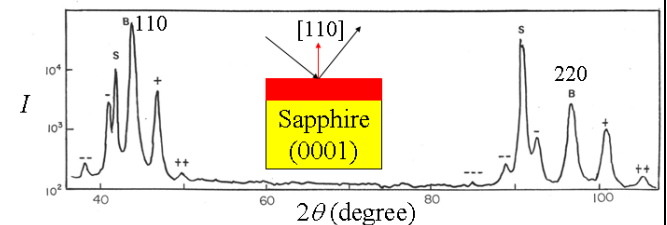
Then the structure factor of the block: $(\Delta d = d_2 - d_1)$

$$F' \propto F_1 + F_1 e^{i\Delta k d_1} + F_1 e^{i\Delta k 2d_1} + \dots + F_1 e^{i\Delta k (n_1 - 1)d_1} + F_2 e^{i\Delta k (n_1 d_1 + \frac{1}{2} \Delta d)} + F_2 e^{i\Delta k (n_1 d_1 + \frac{1}{2} \Delta d + d_2)} + F_2 e^{i\Delta k (n_1 d_1 + \frac{1}{2} \Delta d + 2d_2)} + \dots + F_2 e^{i\Delta k (n_1 d_1 + \frac{1}{2} \Delta d + (n_2 - 1)d_2)}$$

$$F' \propto F_1 \sum_{p=0}^{n_1-1} e^{ip\Delta k d_1} + F_2 e^{i\Delta k (n_1 d_1 + \frac{1}{2} \Delta d)} \sum_{p=0}^{n_2-1} e^{ip\Delta k d_2} = \frac{\sin(n_1 \Delta k d_1 / 2)}{\sin(\Delta k d_1 / 2)} e^{i(n_1 - 1)\Delta k d_1 / 2} \frac{\sin(n_2 \Delta k d_2 / 2)}{\sin(\Delta k d_2 / 2)} e^{i(n_2 - 1)\Delta k d_2 / 2}$$

$|F'|^2$ depends on $F_1 - F_2$ and Δd . See problem set 3.

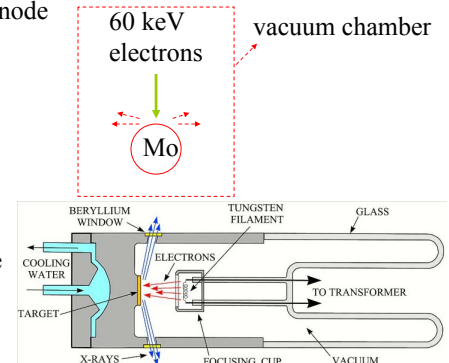
e.g. $\text{V}_8\text{Fe}_8(110)$ superlattice on $\text{Al}_2\text{O}_3(0001)$



(III) Equipment:

(A) X-ray sources

- Synchrotron: very powerful (<http://www-ssrl.slac.stanford.edu>)
- Rotating anode (< 18 kW, 60 kV)



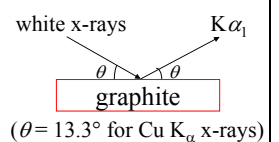
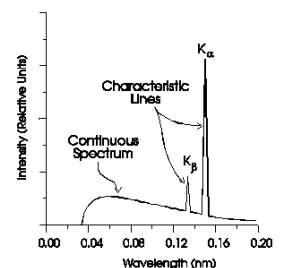
- X-ray tube (< 3 kW, 60 kV)

(B) Monochromators

The x-ray sources generate a white spectrum of x-rays.

Monochromators are used to select only one wavelength for XRD experiment.

- Filter: e.g. a Ni foil can be used to roughly select $K\alpha$ line of a Cu source.
- Single crystal monochromator: a graphite (or germanium, Ge) crystal can be used to select $K\alpha_1$ line of an x-ray source by means of diffraction effect. (Similar to using a grating to select monochromatic light for optical spectrometer.)



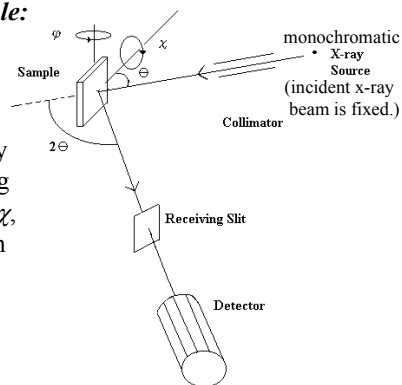
-
- (<http://www-lam.stanford.edu/xlab/Main.htm>)

Simple detector: 1. gas proportional counter
2. scintillation counter

1D detector: curved position-sensitive detectors

2D detector: 1. multiwire proportional counter
2. CCD area detector

- (E) Diffractometer techniques**
4 circle x-ray diffractometer with a simple detector for single crystal sample:



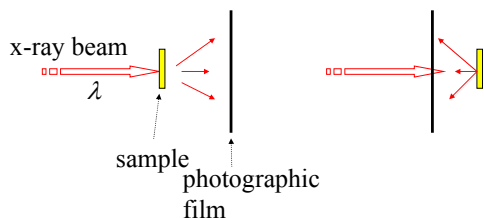
Look for all possible diffractions (that satisfy Bragg law) by adjusting sample orientation (φ, χ, θ) and detector position (2θ). Based on these recorded ($\varphi, \chi, \theta, 2\theta$) data, the crystal structure of the sample can be determined.

4-Circle Goniometer (Eulerian or Kanna Geometry)

(<http://www-structure.llnl.gov/Xray/101index.html>)

Laue diffraction:

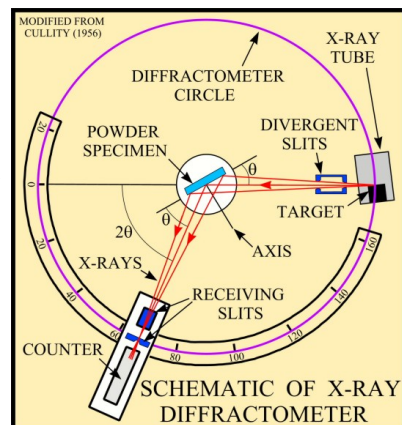
reflection mode



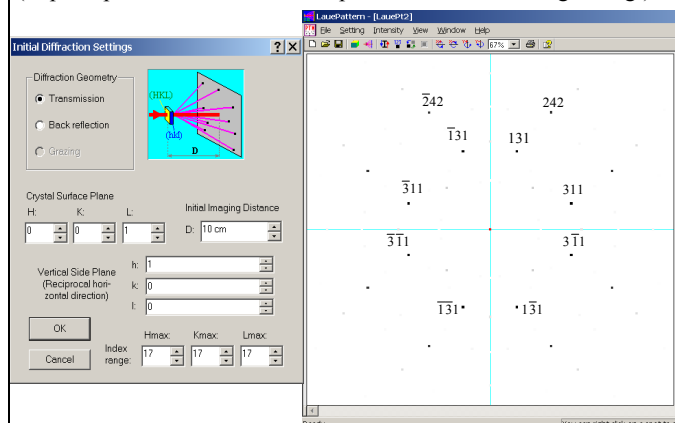
For polycrystalline sample, monochromatic x-rays are used.
For single crystal sample, white x-rays are used.
Main features: very cheap and useful

Incident x-ray beam is fixed and monochromatic (produced by a filter or crystal monochromator).

The sample is rotated by θ and detector is rotated by 2θ . Diffraction beams are observed whenever Bragg law is satisfied.



(<http://ccp14.minerals.csiro.au/ccp/web-mirrors/xianrong-huang/>)



X-ray tube

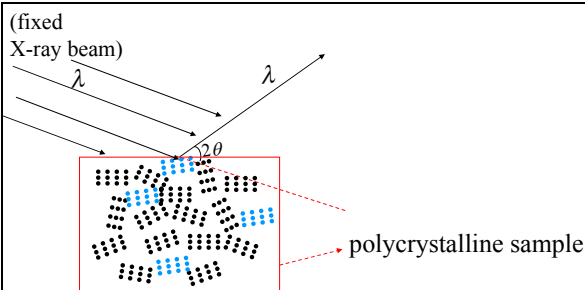
sample

incident (fixed)
x-ray beam

diffraction beams

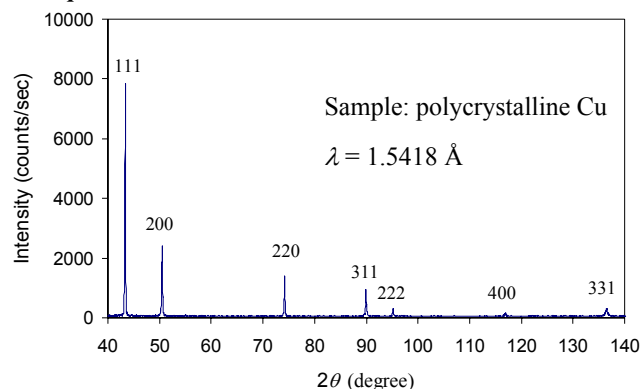
sample

(<http://www.radicon.xraysite.com/index.html>)



For a given material, the 2θ angles that satisfy Bragg equation are known or fixed (see example on next two pages). Only those **grains** in the sample with the right orientation can diffract the x-ray beam.

XRD pattern:



Crystal structure and the unit cell content can be determined using careful analysis of the “powder XRD spectrum”.

ICDD tabulated possible diffractions for many materials: (<http://www.icdd.com/>)

Wavelength= 1.54184									
04-0836	Cu								
Copper									
Copper, syn									
Rad.: CuKα1	λ: 1.5405	Filter: Ni Beta.M	d-sp:						
Cut off:	Int.: Diffract.	I/Corr.:							
Ref: Swanson, Tatge, Natl. Bur. Stand. (U.S.), Circ. 539, 1, 15 (1955)									
Sys.: Cubic		S.G.: Fm3m (225)							
a: 3.6150	b: 3.6150	c: 3.6150	A: 90	B: 90	C: 90				
Ref: Ibid.									
Dx: 8.936	Dm: 8.950	SS/FOM: F ₀ -89(0112, 8)							
Color: Red Pattern taken at 25 C. Sample from metallurgical laboratory of NBS, Gaithersburg, Maryland, USA. CAS #: 7440-50-8. It had been heated in an H ₂ atmosphere at 300 C. Impurities from 0.001-0.01%, Ag, Al, Si, Fe, Sn. Opaque mineral optical data on specimen from unspecified locality. RSC#-60-65, Disp-Std. VHN100-96-104, Ref: IMA Commission on Ore Microscopy QDF. Measured density and color from Dana's System of Mineralogy, 7th Ed., 199. Cu type. Gold group. gold subgroup. FSC: c24. Mwt: 63.55. Volume[CD]: 47.24.									

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(IV) Useful resources in internet

Basic:

- (1) Ron Jenkins, “X-ray Techniques: Overview” in *Encyclopedia of Analytical Chemistry*, ed. by R.A. Meyers, pp. 13269–13288 (Wiley 2000)
- (2) **x-ray data booklet** (for x-ray data) (<http://xdb.lbl.gov/>)

Tutorial:

- (1) Teaching guide X-Ray and neutron diffraction (<http://www.kri.physik.uni-muenchen.de/geo/crystal/teaching/content.html>)
- (2) <http://www.matscieng.sunysb.edu/esm512/webbers-calcs/>
- (3) <http://www.mrl.ucsb.edu/mrl/centralfacilities/xray/xray-basics/Xray-basics.html>