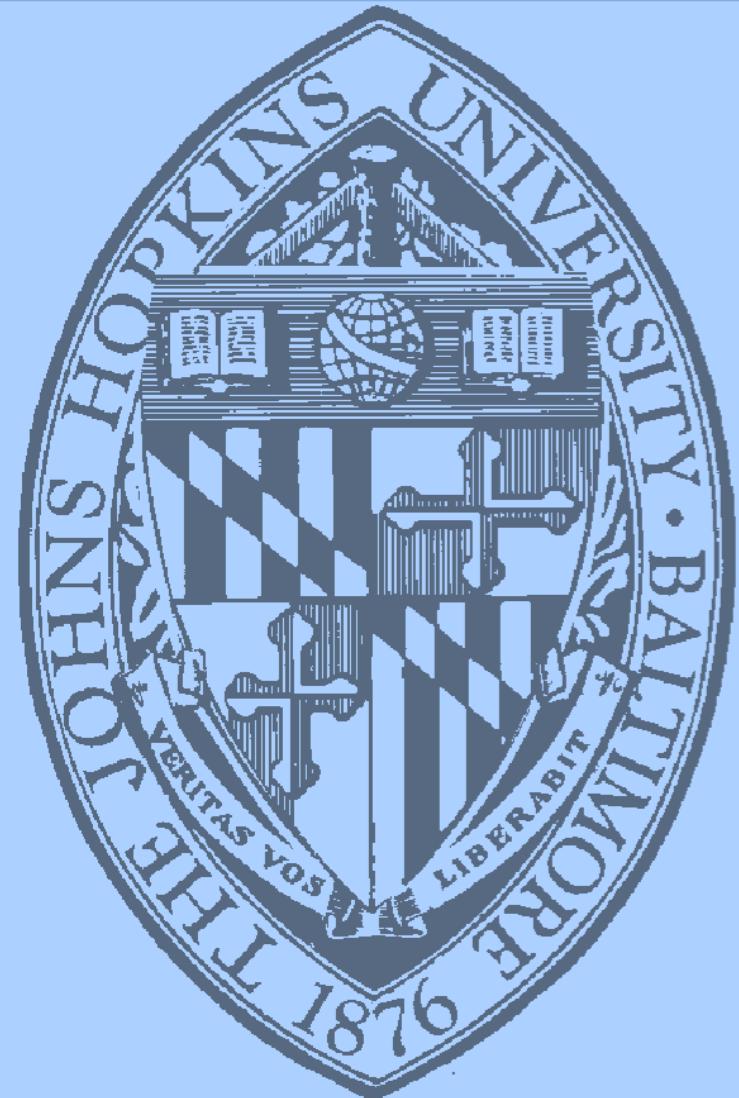


# Plastic Deformation in Amorphous Materials

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***Michael L. Falk***  
***Materials Science and Engineering***  
***Whiting School of Engineering***  
***Johns Hopkins University***



# My Thread This Week

**Mon**  
Intro to  
Mechanics  
of Materials

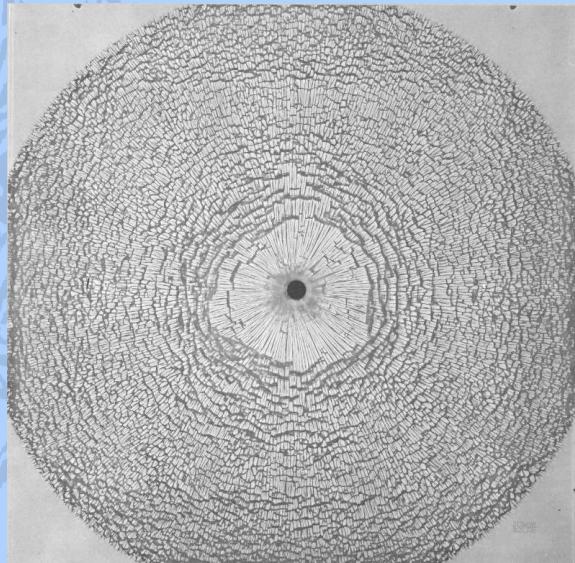
**Tues**  
Intro to  
Molecular  
Dynamics

**Wed**  
Plasticity in  
Amorphous  
Materials

**Thurs**  
STZ  
Constitutive  
Theory

# Amorphous Materials (Glass)

- Represent a broad class of useful metal, ceramic and polymer materials in current application and in development.  
Their common characteristic is a lack of crystalline order.



Safety Glass Cracking,  
Harold Edgerton, 1938

10 Sept 08

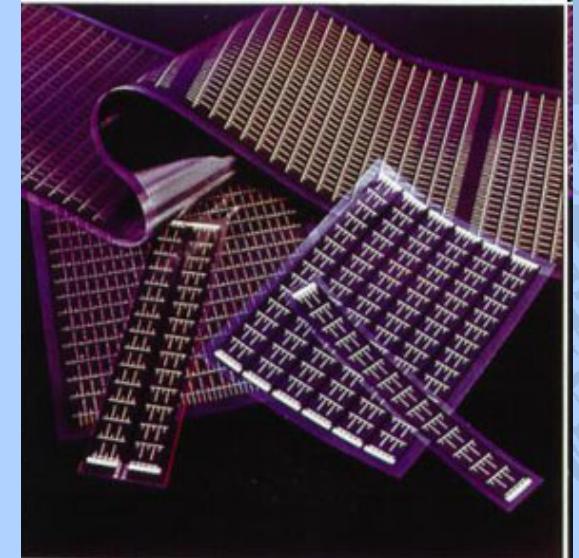


Metallic Glass Golf Club  
Head, Liquidmetal Tech.

CUHK-ITP Mini Workshop

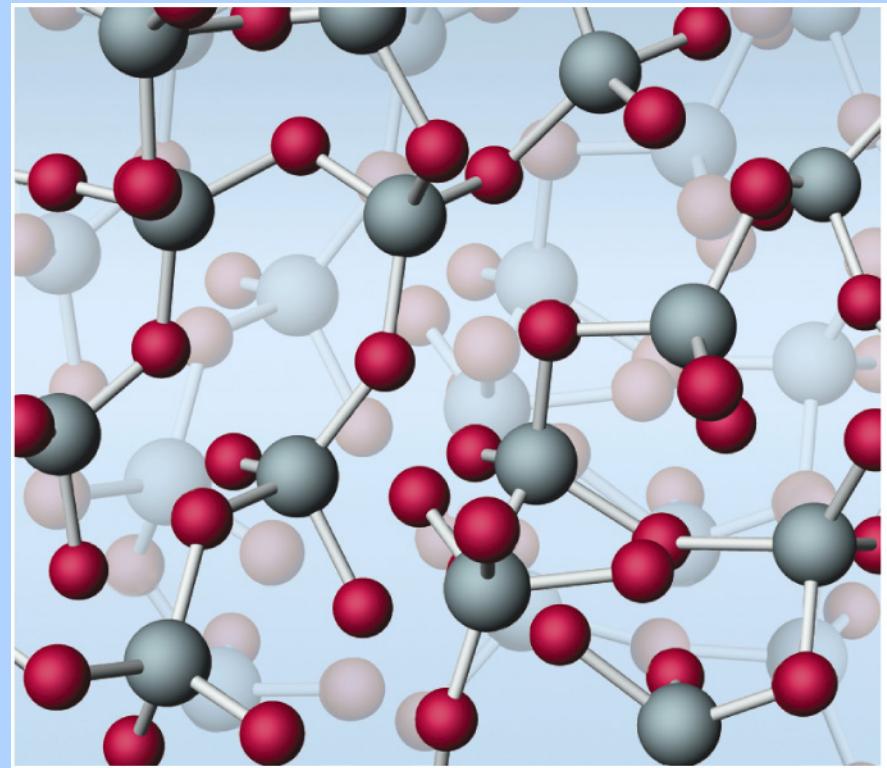
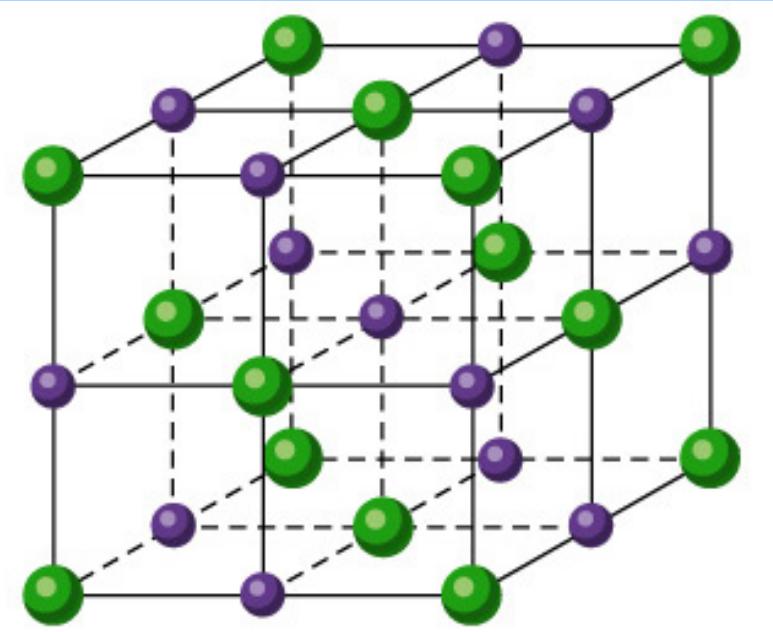


PMMA house wares



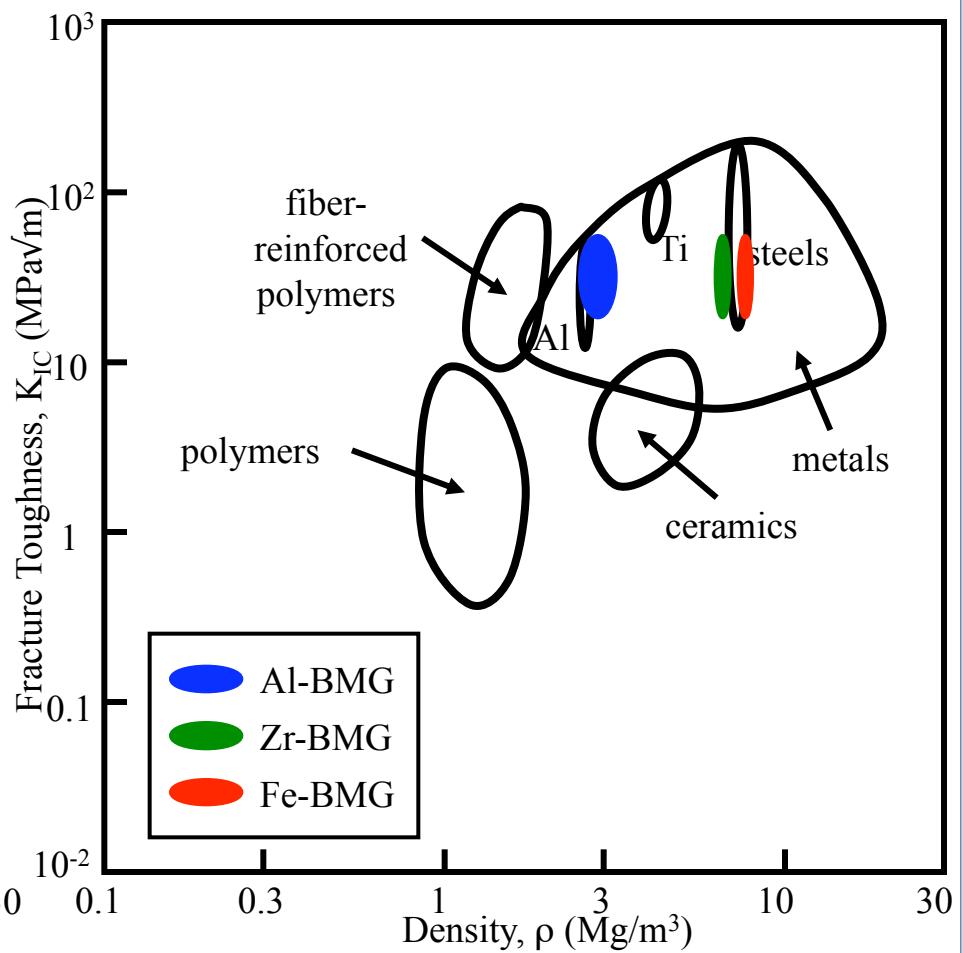
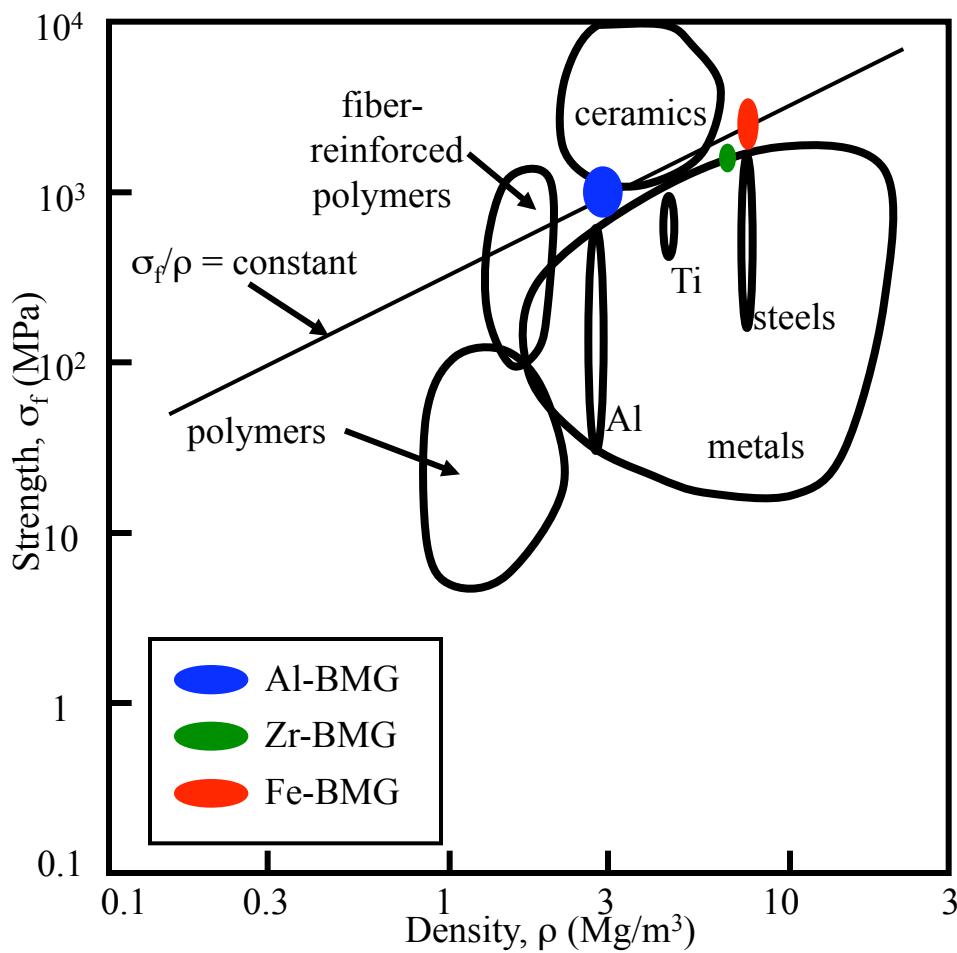
Flexible Si Solar Cells,  
Iowa Thin Film Tech.

# Orderly Crystals vs. Disorderly Glass



# Metallic Glass

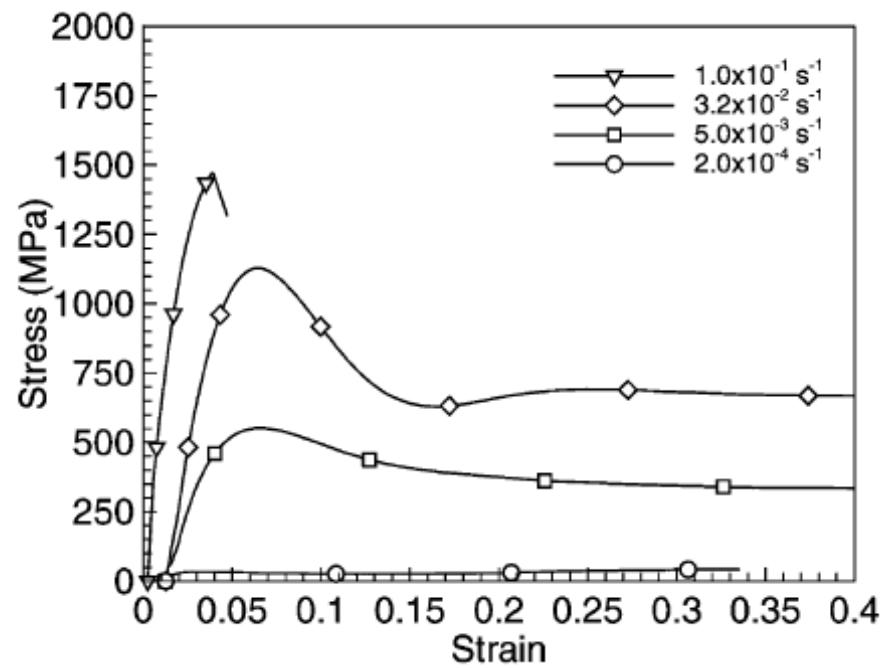
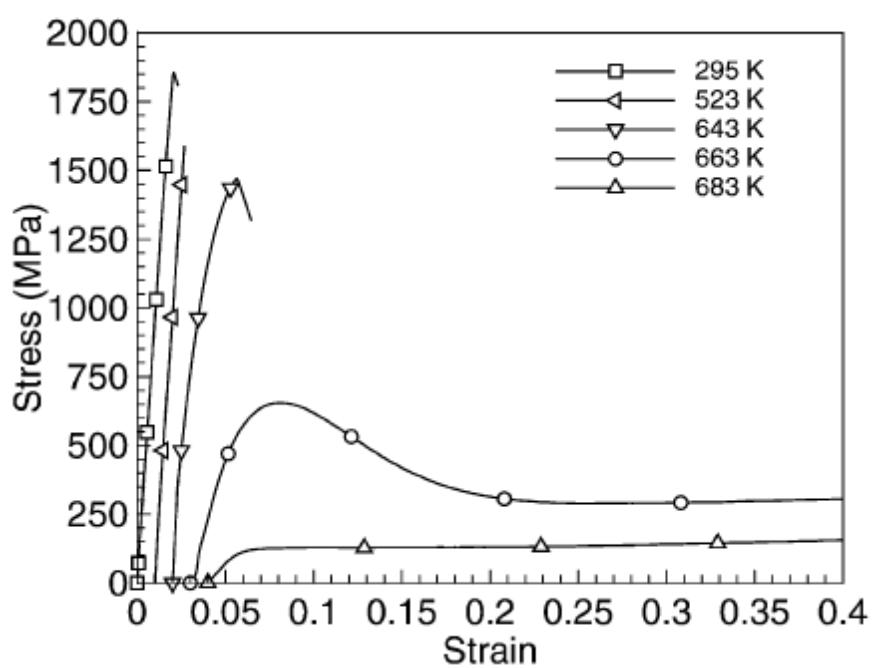
graphs courtesy of Katherine Flores, OSU



# Applications of Bulk Metallic Glasses

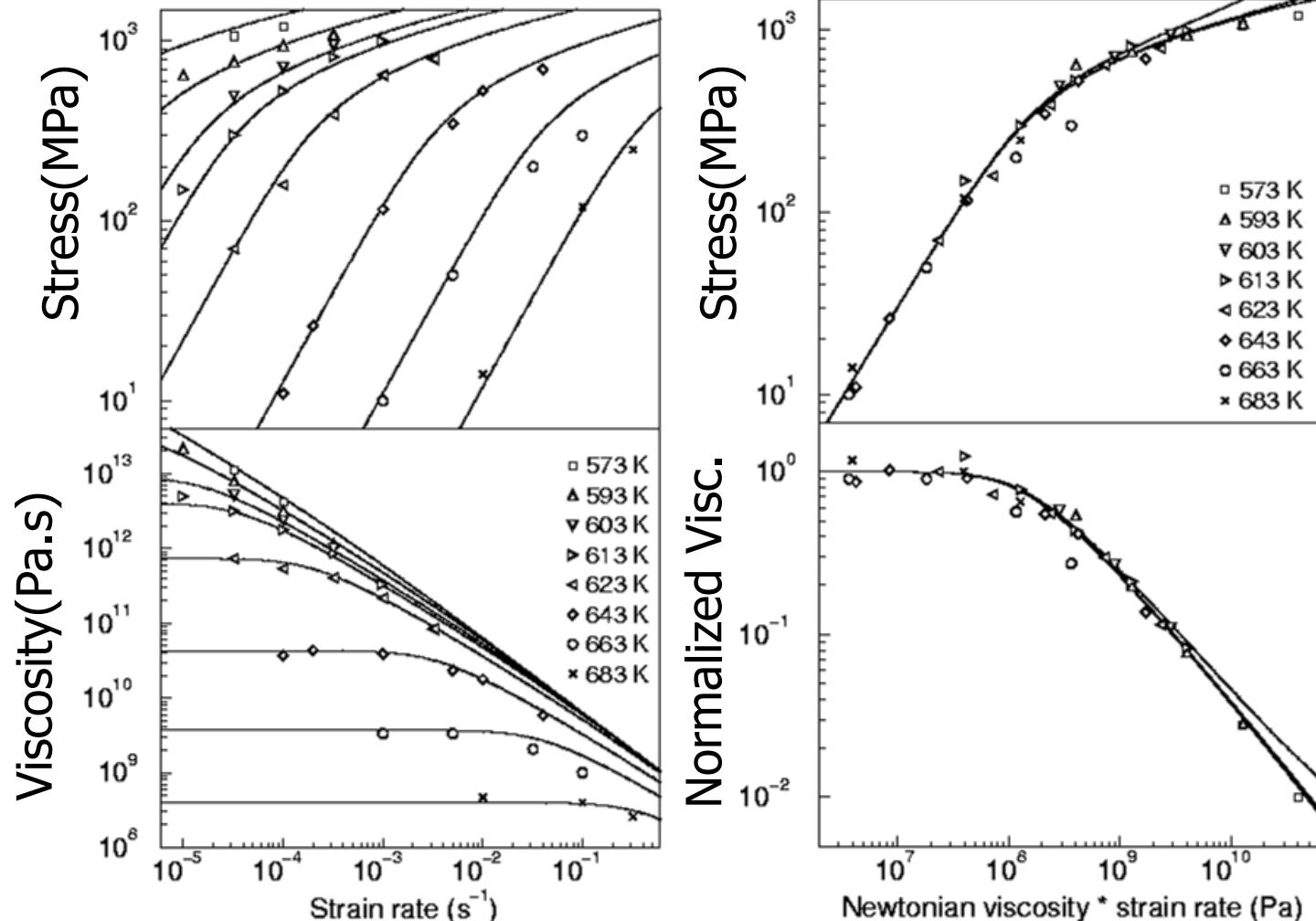


# Mechanical Response Near $T_g$



Lu, Ravichandran, Johnson Acta Mat. 51, 3429 (2003)

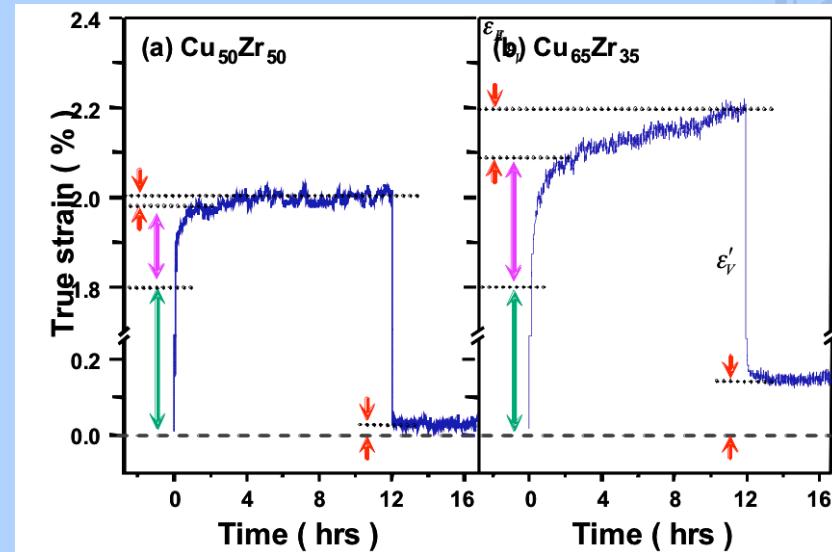
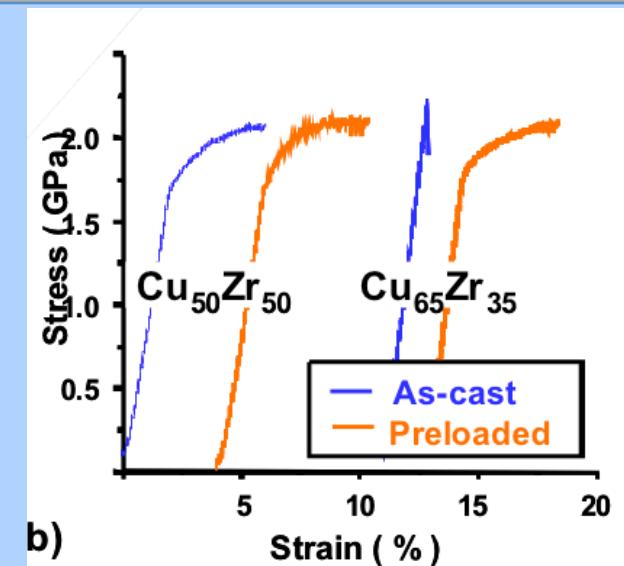
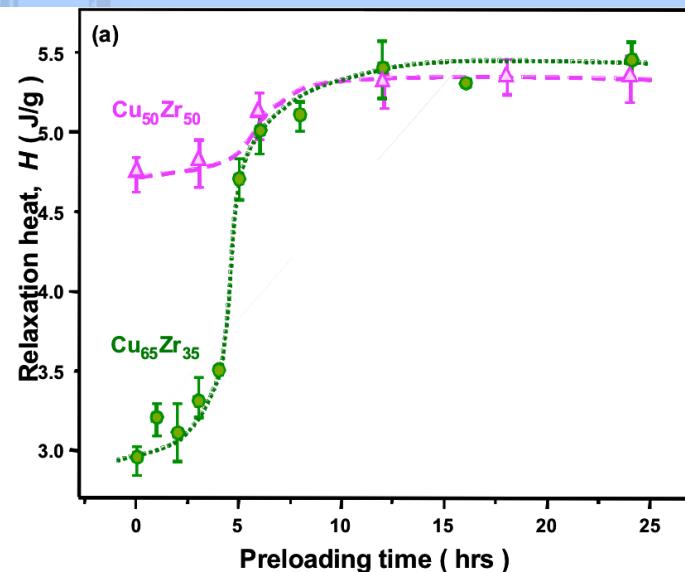
# Rheology Near $T_g$



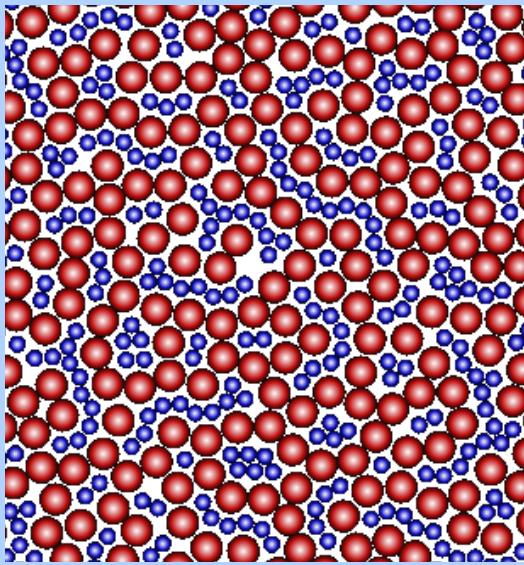
Lu, Ravichandran, Johnson Acta Mat. 51, 3429 (2003)

# Behavior at Room Temperature

- Recent work in collaboration with Jae-Chul Lee at Korea University shows apparent contradiction
- Alloys that exhibit a large degree of plasticity when loaded at constant stress, show low ductility loaded at constant strain rate
- This appears to be related to structural changes during shear

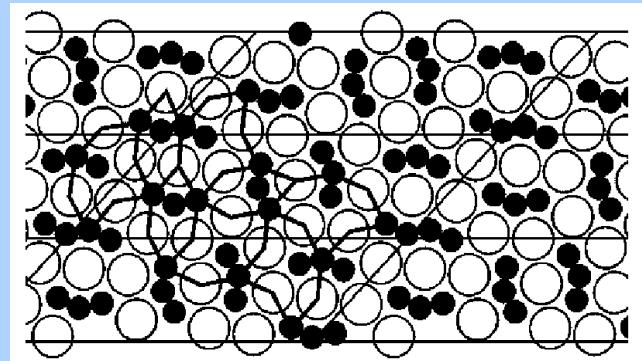


# 2D Simulation System

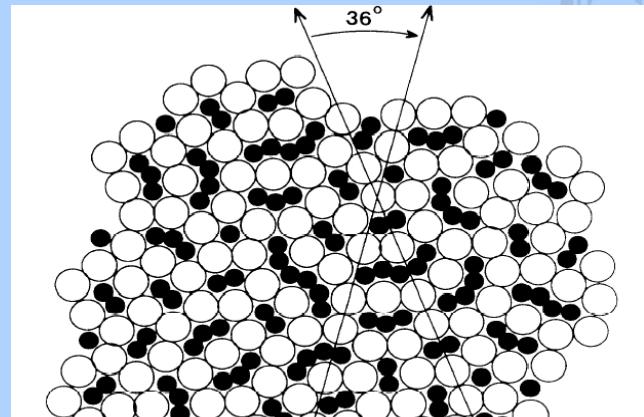


(Lancon et al, Europhys. Lett, 1986)

- 2D binary Lennard-Jones 12-6 potential
  - Binary system with quasi-crystalline packing
- 45:55 composition, 20,000-80,000 atoms
- $T_{MCT} \approx 0.325$

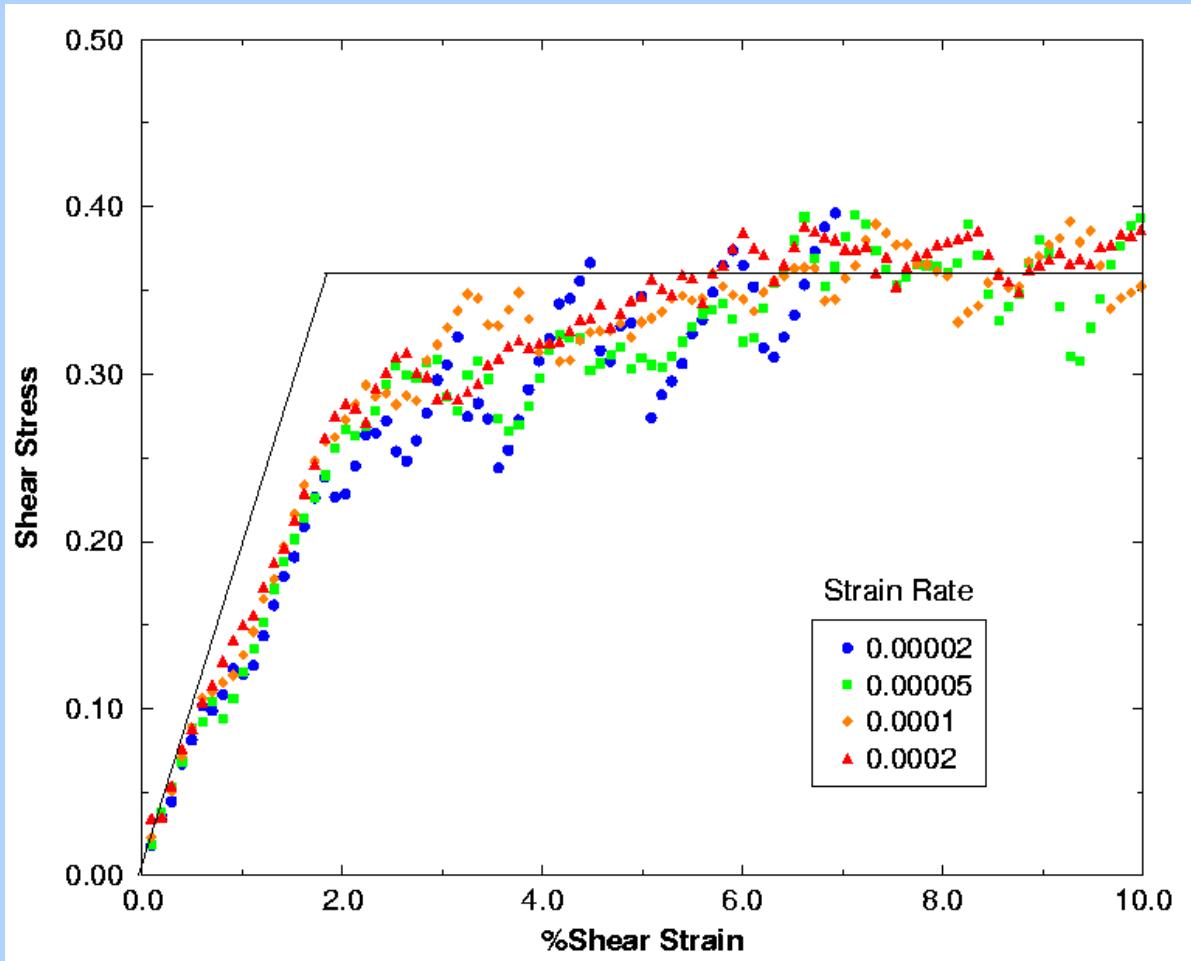


Lee, Swendsen, Widom (2001)



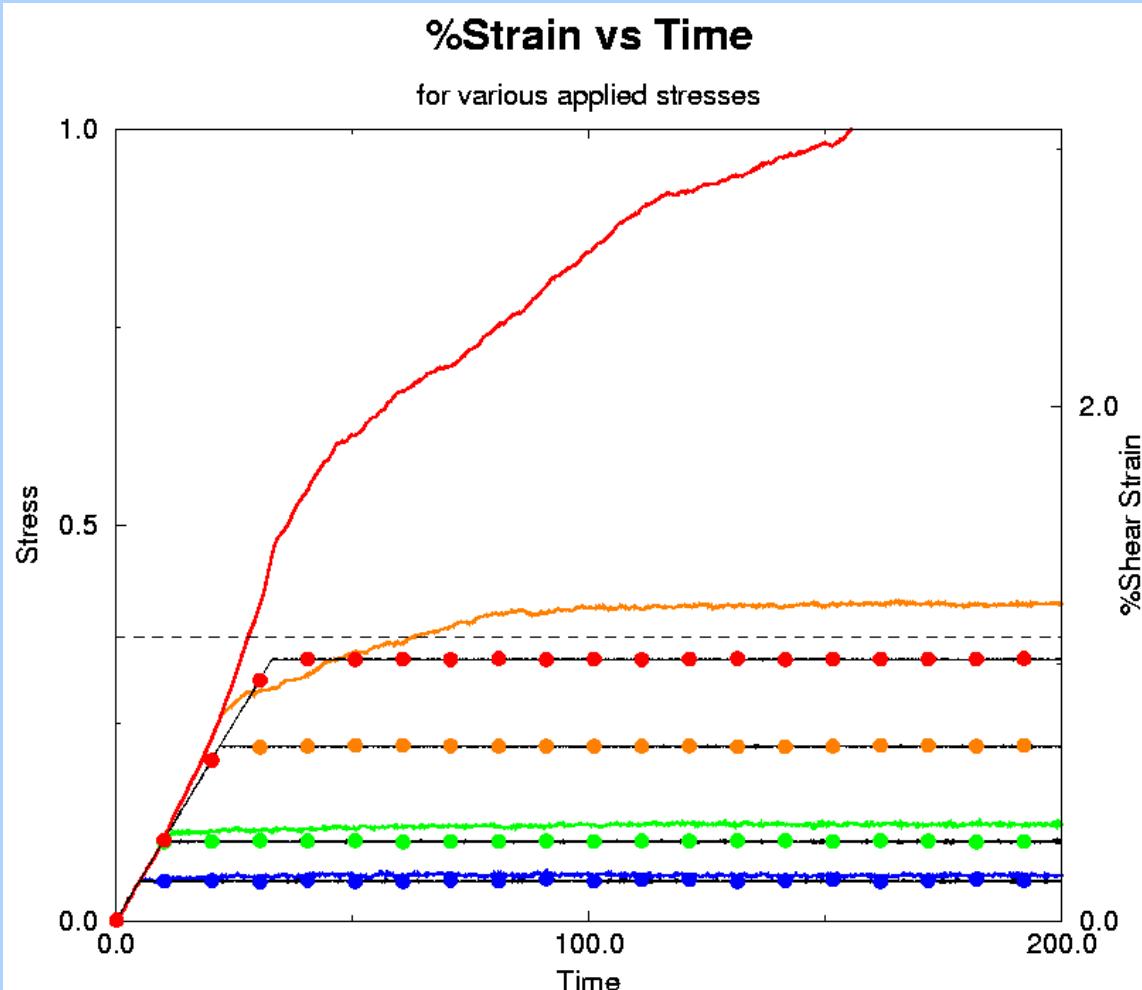
Widom, Strandburg, Swendsen (1987)

# Homogeneous MD Simulations



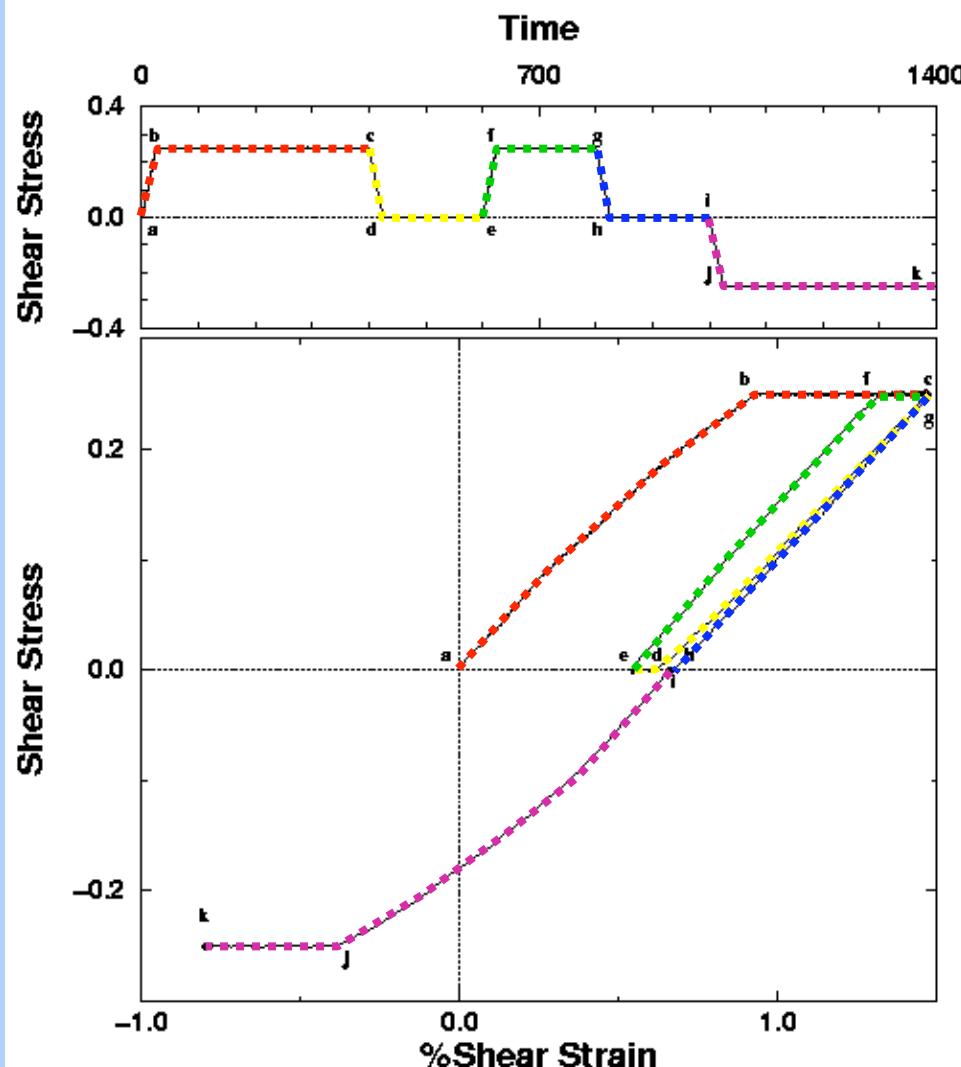
MLF and JS Langer, PRE, Vol. 57, pp. 7192-7205 (1998)

# Homogeneous MD Simulations



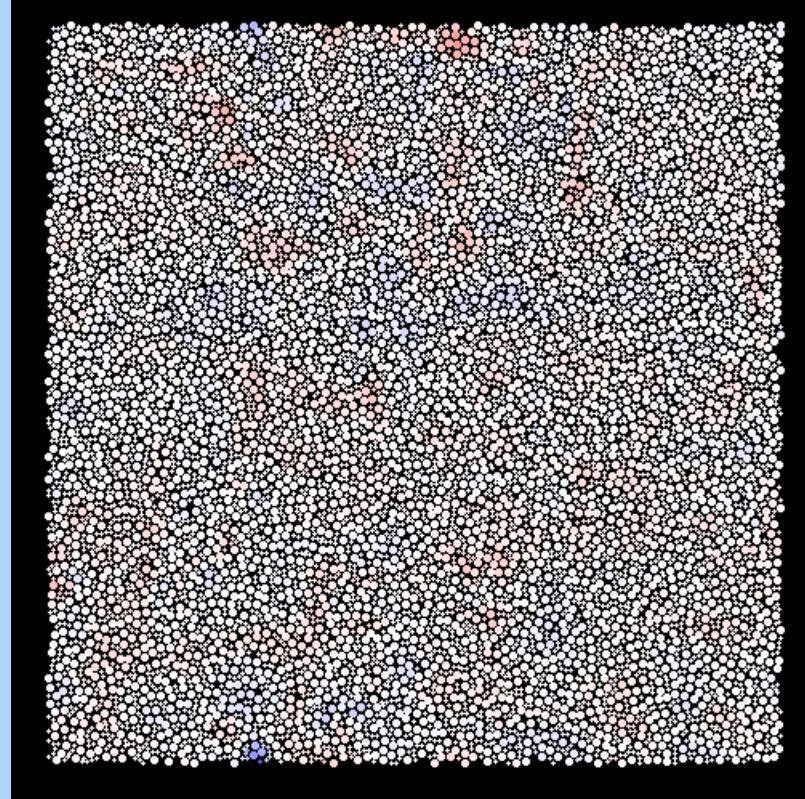
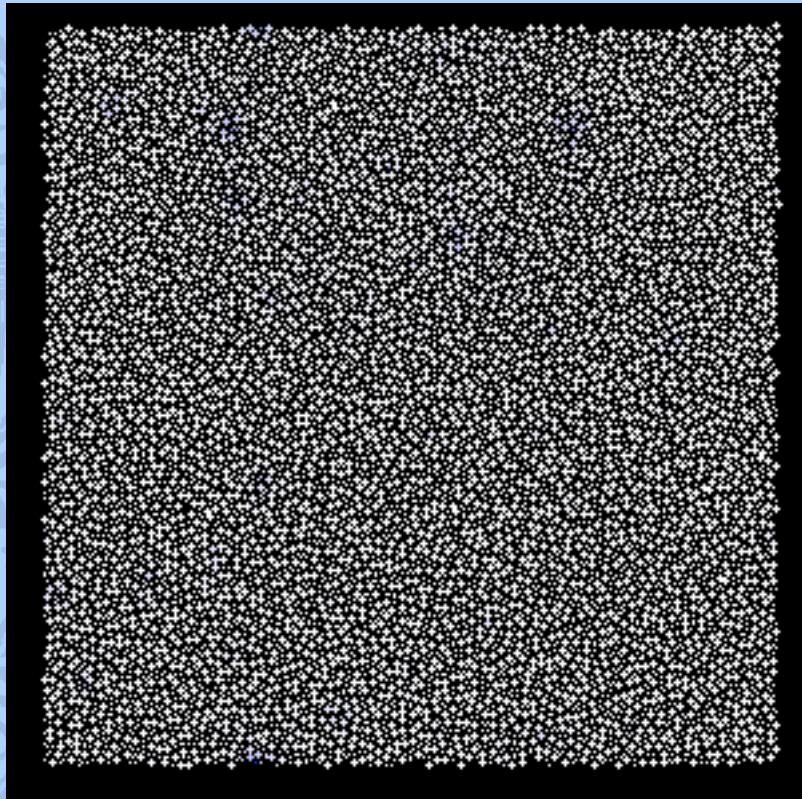
MLF and JS Langer, PRE, Vol. 57, pp. 7192-7205 (1998)

# Homogeneous MD Simulations

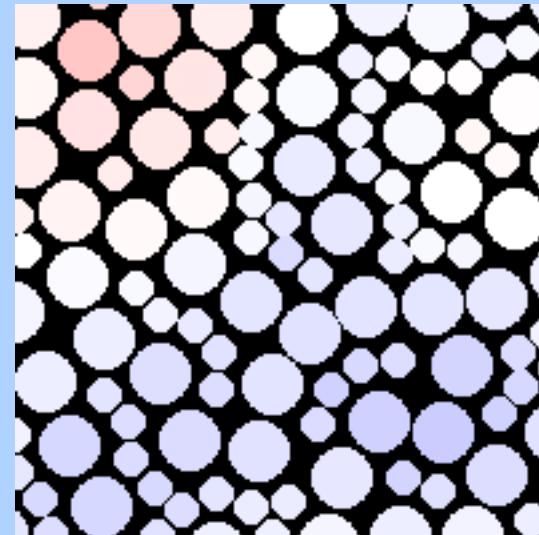
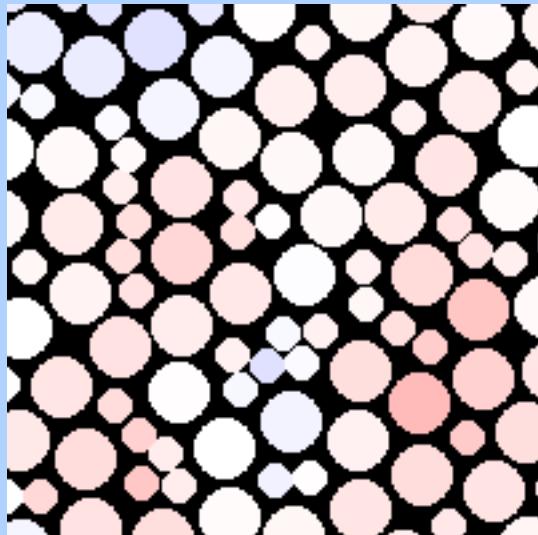


MLF and JS Langer, PRE, Vol. 57, pp. 7192-7205 (1998)

# STZ Picture



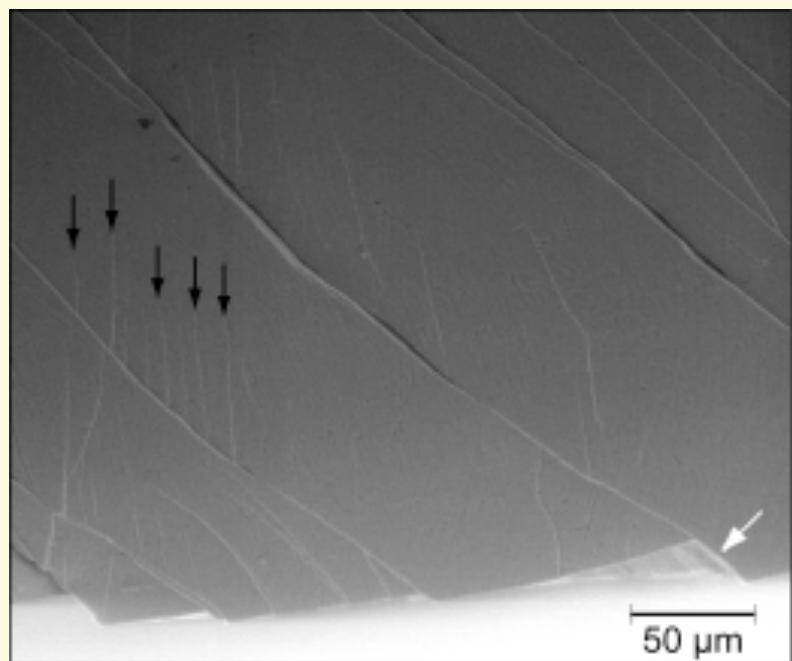
# STZ Picture



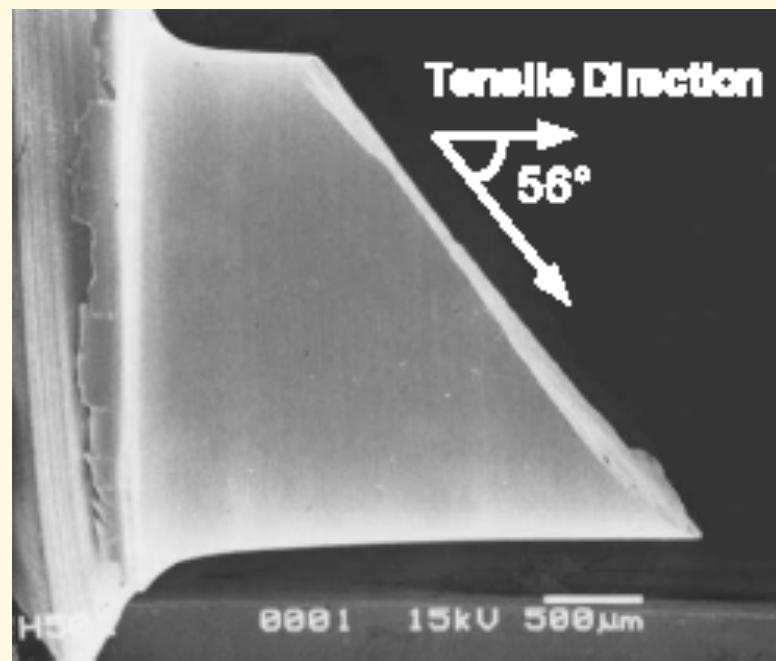
- STZs have a particular orientation. They are susceptible to shear to the extent that the shear is along this direction.
- STZs are reversible until their environment rearranges. They behave as 2-state systems.
- STZs are transient. They can be created and destroyed by neighboring plastic activity.

# Metallic Glass Failure

**strain localization (shear banding) is the primary failure mode**

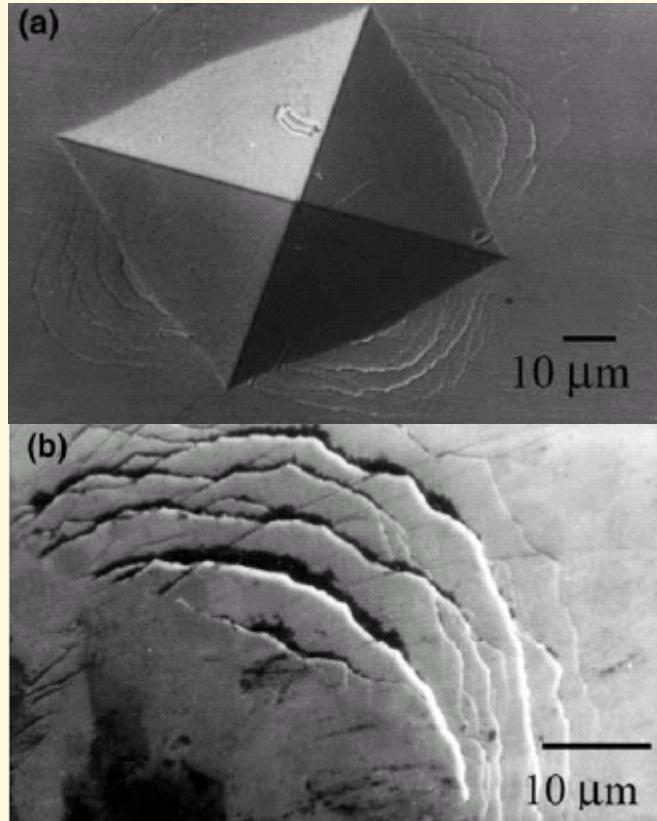


Electron Micrograph of Shear Bands  
Formed in Bending Metallic Glass  
Hufnagel, El-Deiry, Vinci (2000)

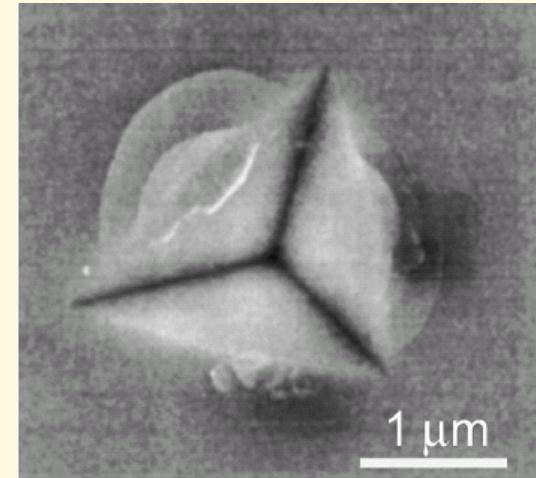


Quasistatic Fracture Specimen  
Mukai, Nieh, Kawamura, Inoue,  
Higashi (2002)

# Indentation Testing of Metallic Glass



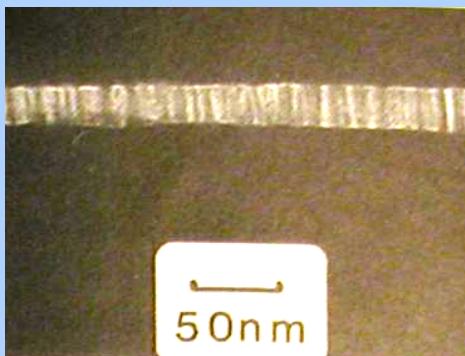
**"Hardness and plastic deformation in a bulk metallic glass"**  
Acta Materialia (2005)  
U. Ramamurtty, S. Jana, Y. Kawamura, K. Chattopadhyay



**"Nanoindentation studies of shear banding in fully amorphous and partially devitrified metallic alloys"**  
Mat. Sci. Eng. A (2005)  
A.L. Greer, A. Castellero, S.V. Madge, I.T. Walker, J.R. Wilde

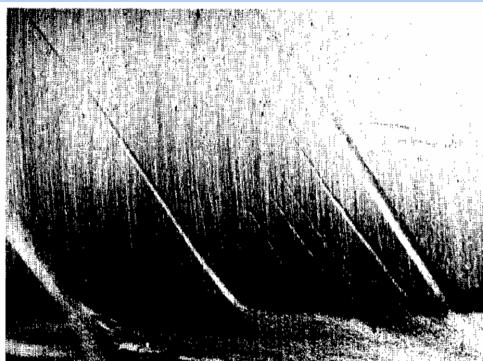
# Shear Bands

Polymer Crazing



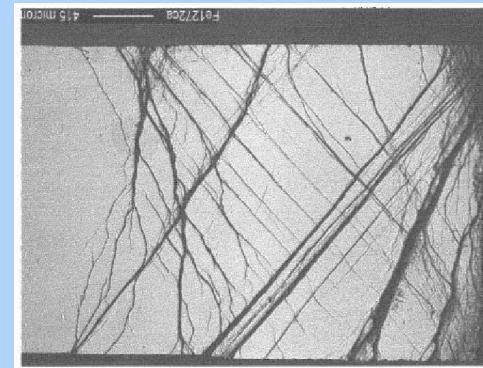
*Young and Lovell (1991)*

Mild Steel



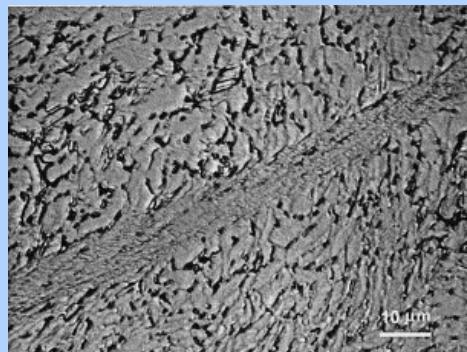
*Van Rooyen (1970)*

Nanogranulated Metal



*Wei, Jia, Ramesh  
and Ma (2002)*

Steel @ High Rate



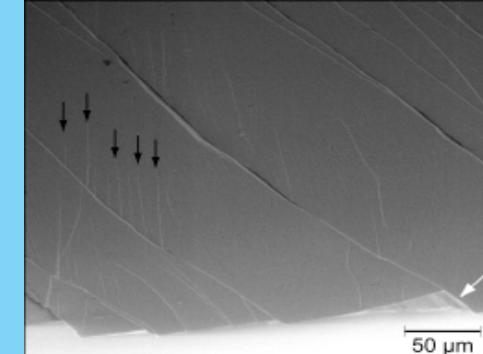
*Xue, Meyers  
and Nesterenko (1991)*

Granular Materials



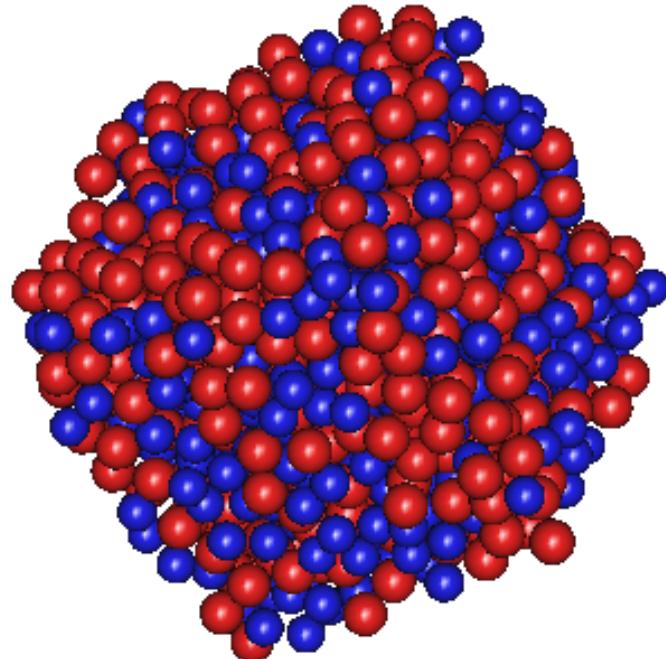
*Mueth, Debreegas  
and et. al. (2000)*

Bulk Metallic Glasses



*Hufnagel, El-Deiry  
and Vinci (2000)*

# Simulated System: 3D Binary Alloy

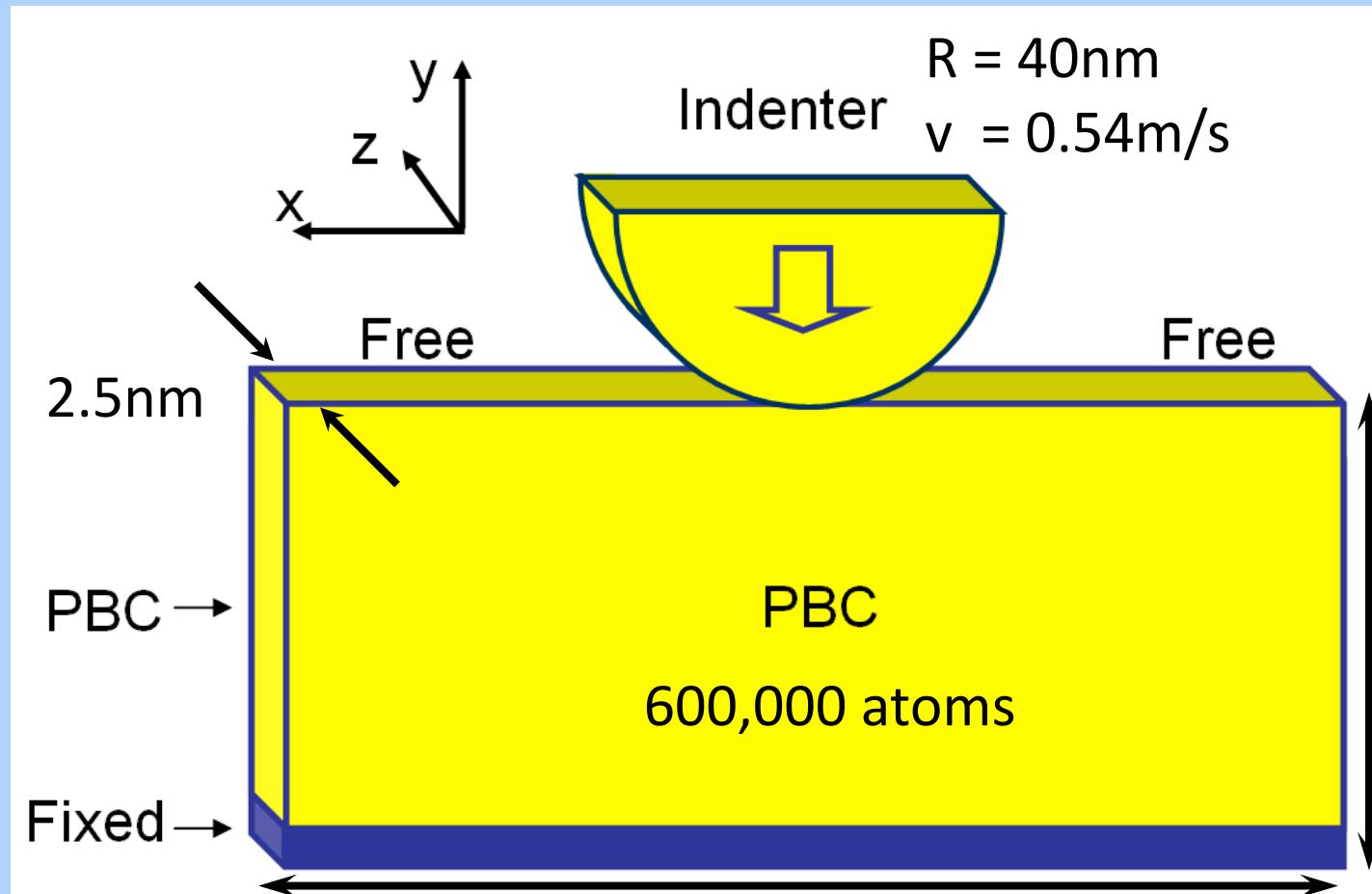


- Wahnstrom Potential (PRA, 1991)
- Rough Approximation of  $\text{Nb}_{50}\text{Ni}_{50}$
- Lennard-Jones Interactions
- Equal Interaction Energies
- Bond Length Ratios:
  - $a_{\text{NiNi}} \sim \frac{5}{6} a_{\text{NbNb}}$
  - $a_{\text{NiNb}} \sim \frac{11}{12} a_{\text{NbNb}}$
- $T_g \sim 1000\text{K}$
- Studied previously in the context of the glass transition (Lacevic, *et. al.* PRB 2002)

- Unlike the simulation of crystalline systems, it is not possible to skip simulating the processing step
- Glasses were created by quenching at 3 different rates: 50K/ps, 1K/ps and 0.02 K/ps

# Metallic Glass Nanoindentation

Simulations performed using molecular dynamics code across 64 nodes of a parallel cluster



Y. Shi, MLF, Acta Materialia, 55, 4317 (2007)

100nm

# Metallic Glass Nanoindentation

color = deviatoric strain

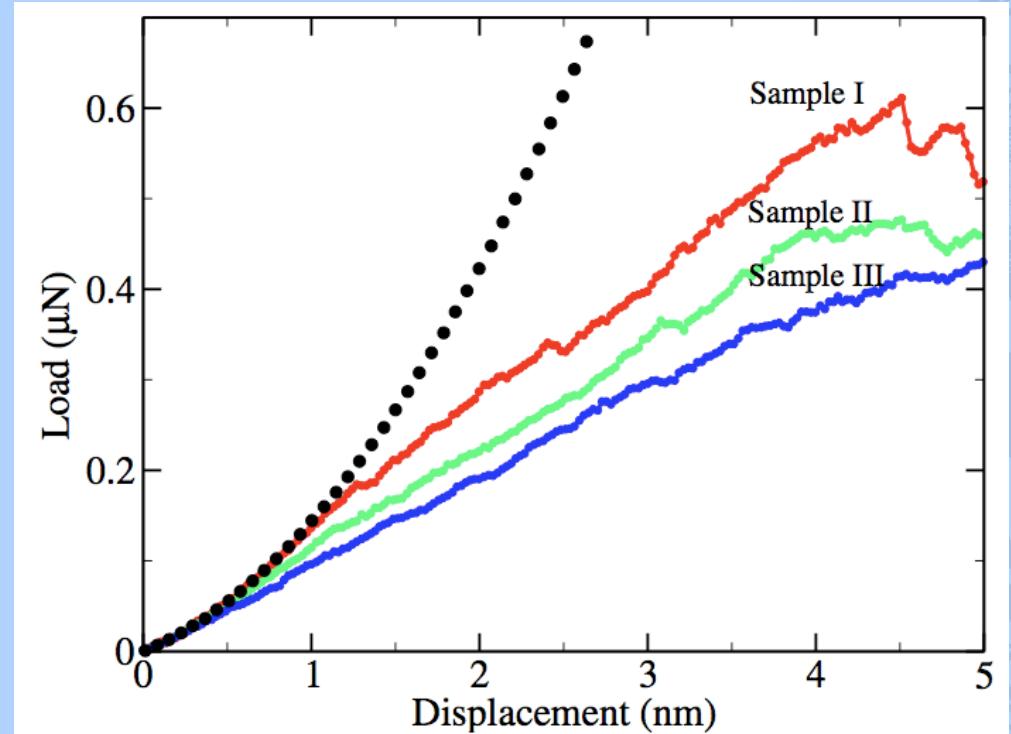
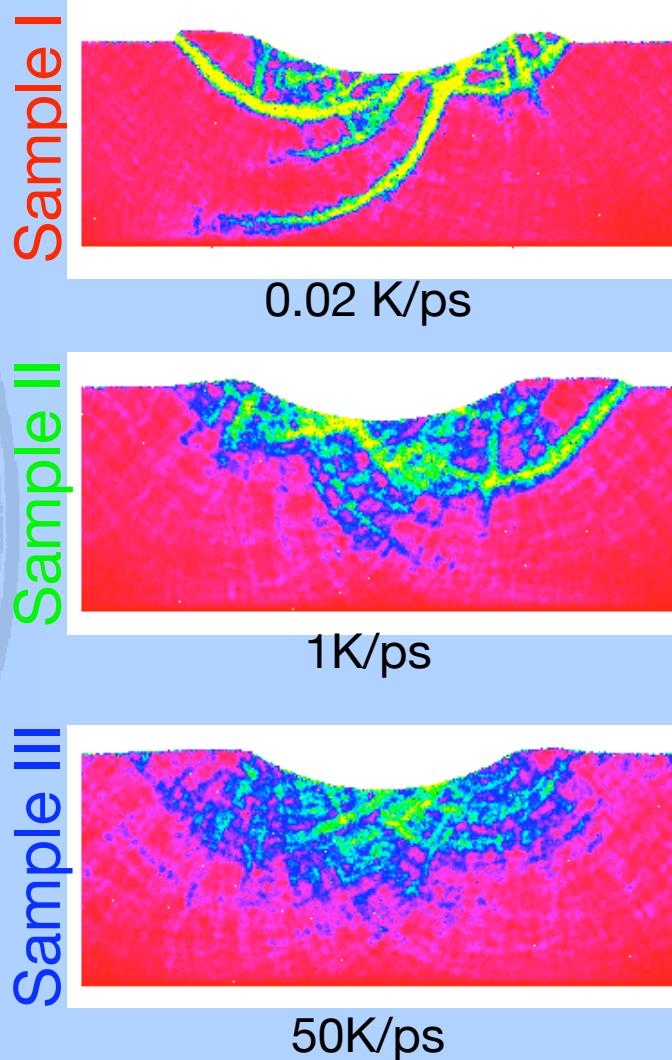
0%

40%



Y. Shi, MLF, Acta Materialia, 55, 4317 (2007)

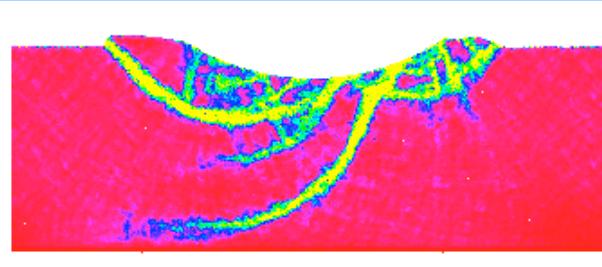
# Metallic Glass Nanoindentation



Y. Shi, MLF, Acta Materialia, 55, 4317 (2007)

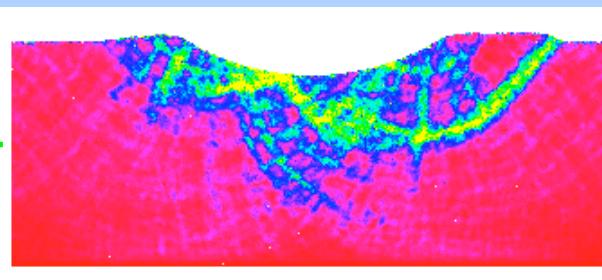
# Metallic Glass Nanoindentation

Sample I



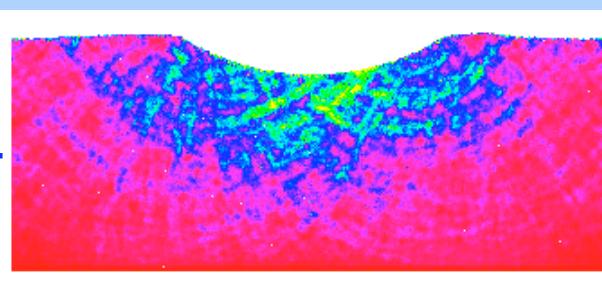
0.02 K/ps

Sample II

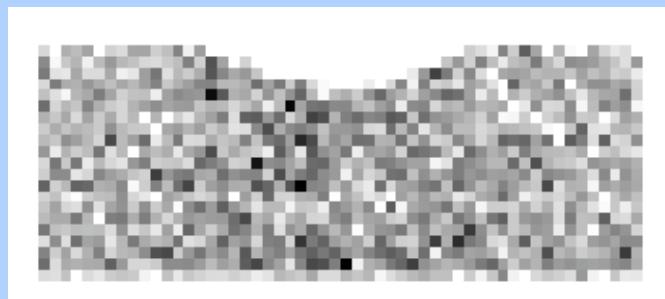
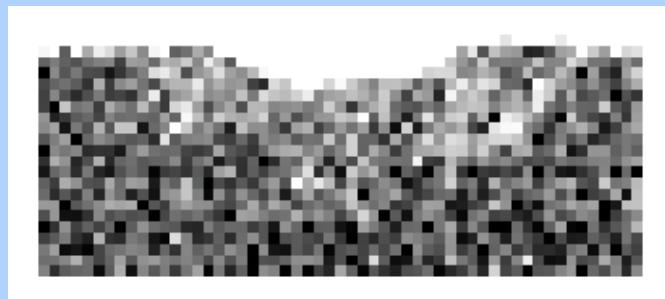


1K/ps

Sample III



50K/ps



**Y. Shi, MLF, Acta Materialia, 55, 4317 (2007)**

CUHK-ITP Mini Workshop

10 Sept 08

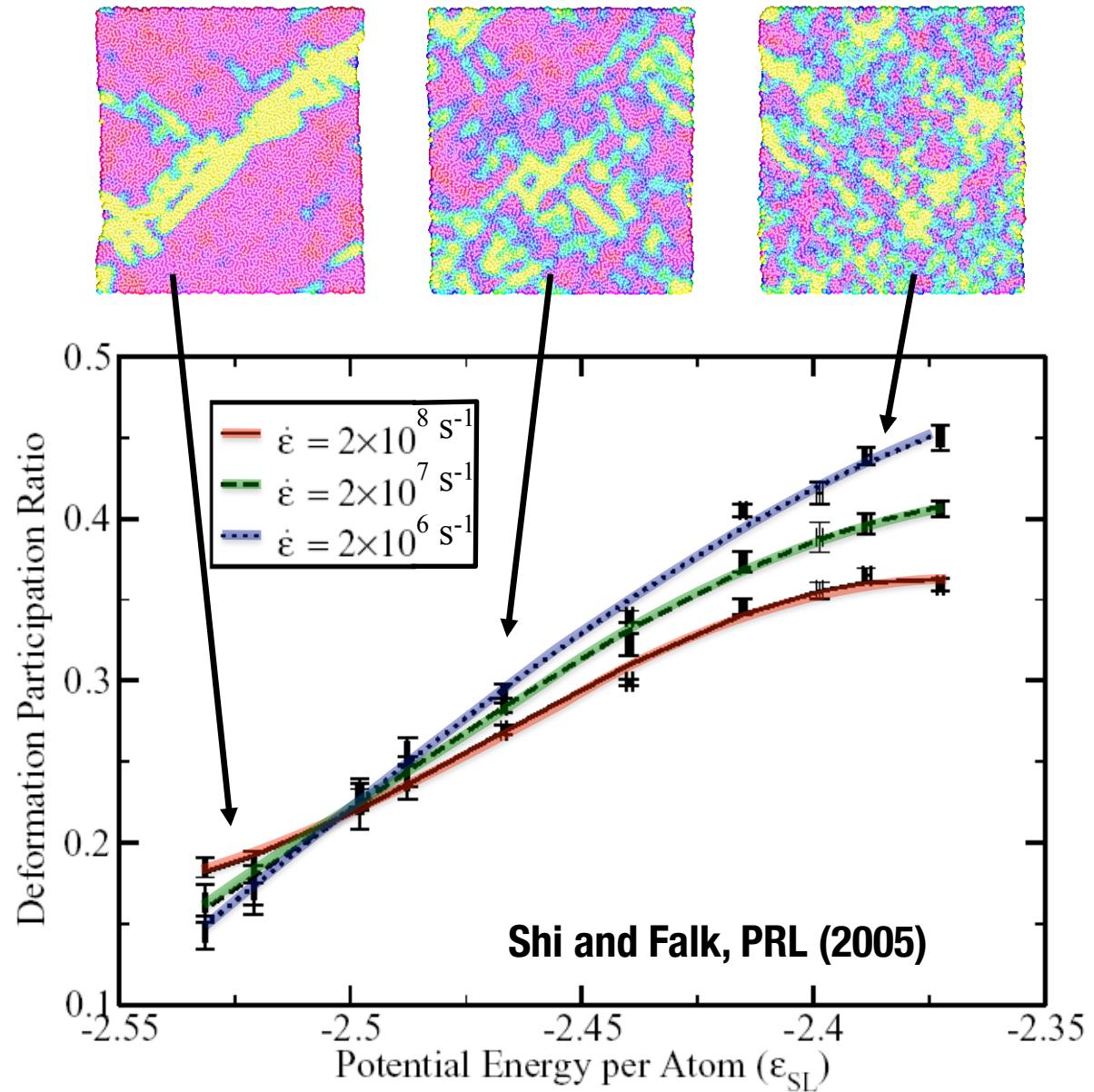
23

# Quantifying the Dependence of Localization on Quench Rate (2D)

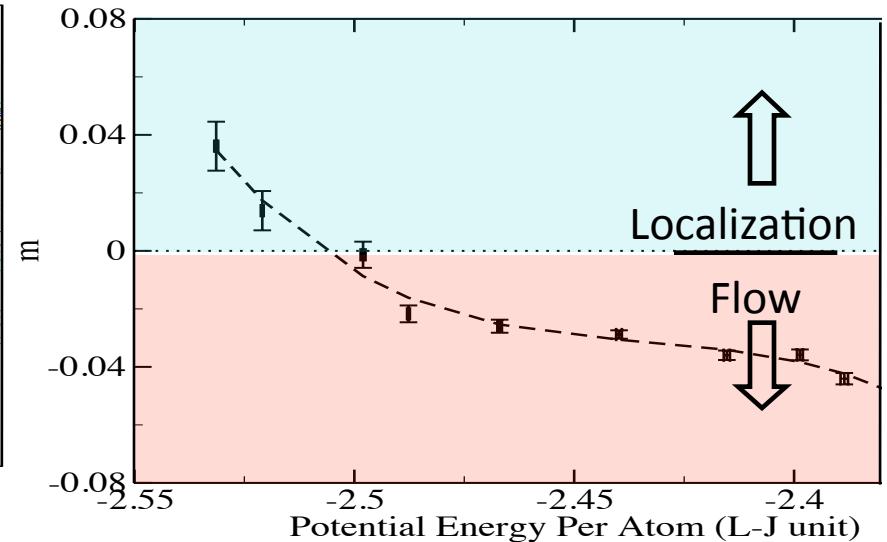
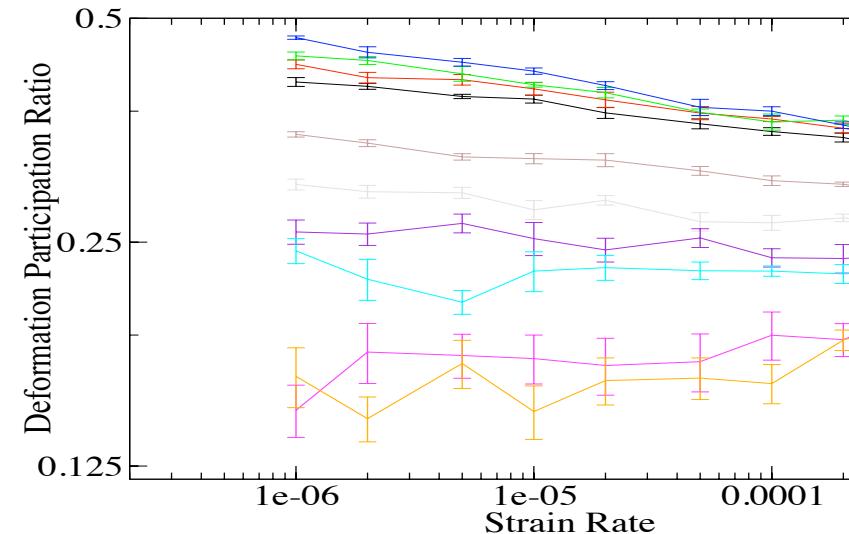
- Performed **756** individual 2D uniaxial tensile test simulations at  $0.1 T_g$
- **10 different quench schedules** starting from equilibrium liquids
- **6-10 samples** at each quench schedule
  - Each of these **84** specimens was tested at **9 different strain rates** spanning **2 orders of magnitude**

# Shear Localization (DPR) vs. PE

- Deformation Participation Ratio: Percentage of material with a local shear strain larger than the nominal strain
- Low strain rate favors homogenous deformation in instantaneously quenched samples
- Low strain rate favors inhomogeneous deformation in gradually quenched samples.



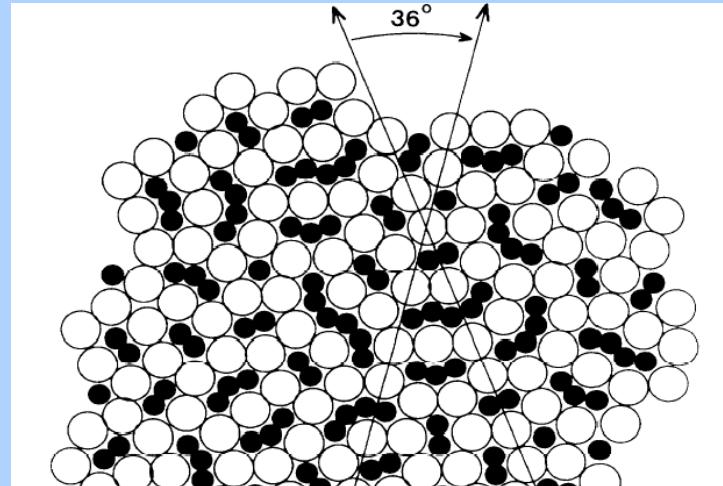
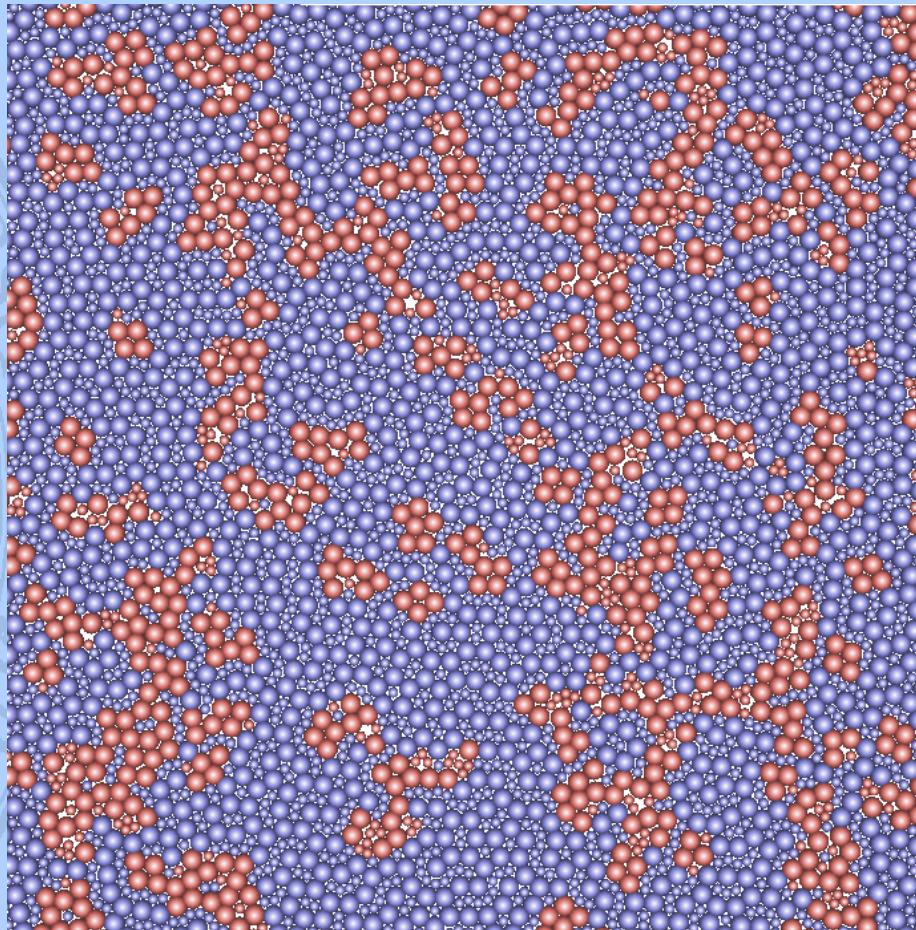
# Strain-rate sensitivity of DPR



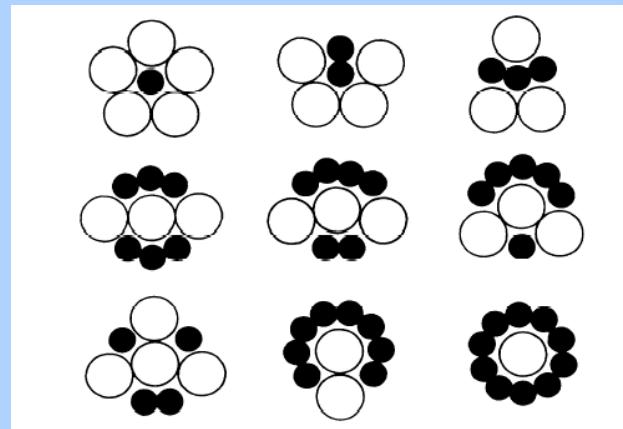
$$DPR \approx A \dot{\varepsilon}^m$$

For  $\dot{\varepsilon} \rightarrow 0$  and system size  $\rightarrow \infty$   
 $m < 0$ : homogenous deformation  
 $m \geq 0$ : localized deformation

# Local Structural Analysis

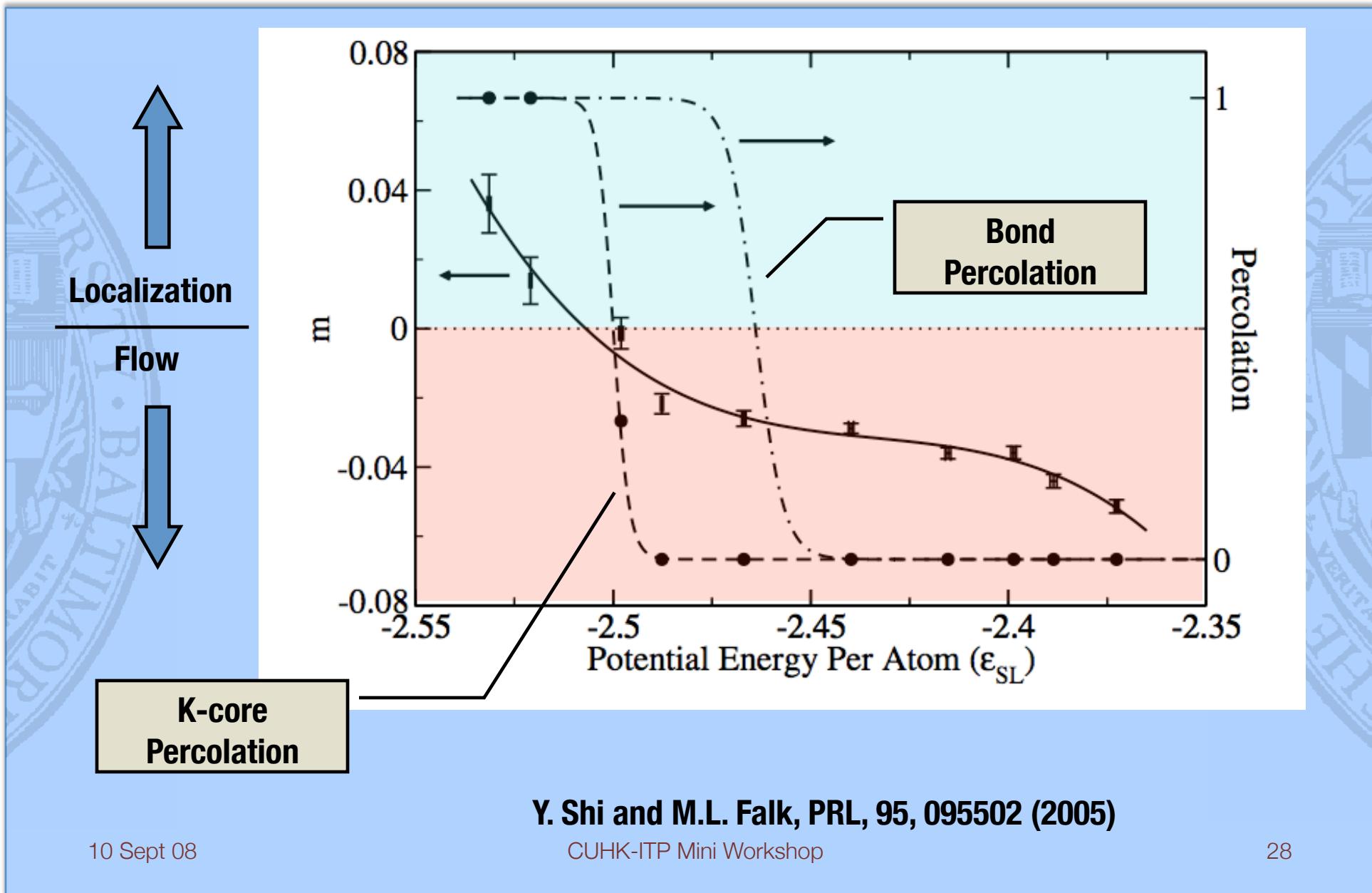


Widom, Strandburg, Swendsen (1987)



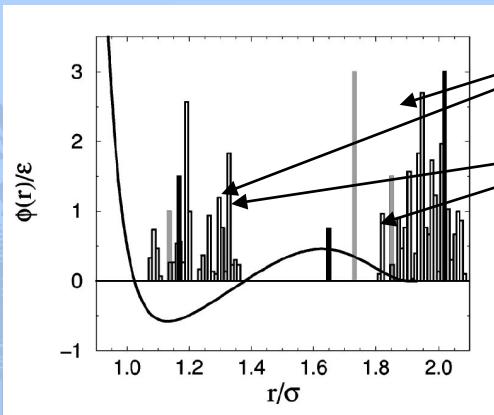
Complete set of low-energy local environments (Widom, 1987)

# Percolation and Localization



# 3D Simulation Potentials

Dzugutov Potential

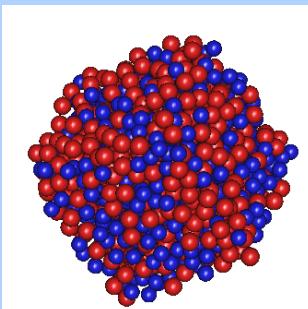


Roth and Denton, PRE (2000)

- 3D Monoatomic
- Energy penalties for crystalline phases
- Dodecagonal quasicrystal
- $T_{MCT} = 0.4$

Zetterling et al., JNCS (2001)

Wahnstrom LJ Binary



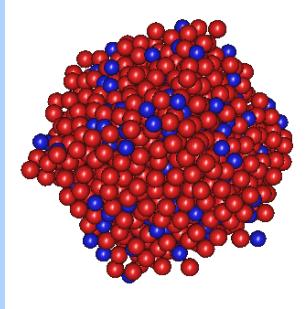
Bond length	Bond strength
AA 1.000	AA 1.0
AB 0.917	AB 1.0
BB 0.833	BB 1.0

- 3D binary LJ 12-6 potential
- 50:50 composition, 144,000 atoms
- $T_{MCT} = 0.57$

Wahnstrom, PRA, 1991

Lacevic et al., PRB, 2002

Kob-Andersen LJ Binary

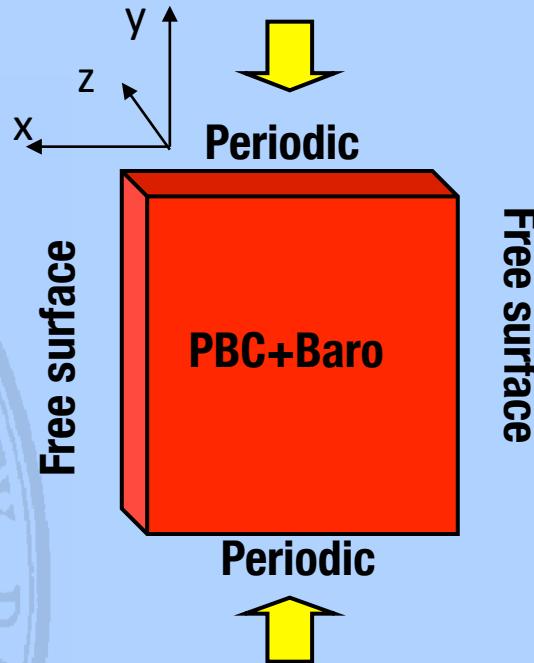


Bond length	Bond strength
AA 1.00	AA 1.0
AB 0.80	AB 1.5
BB 0.88	BB 1.0

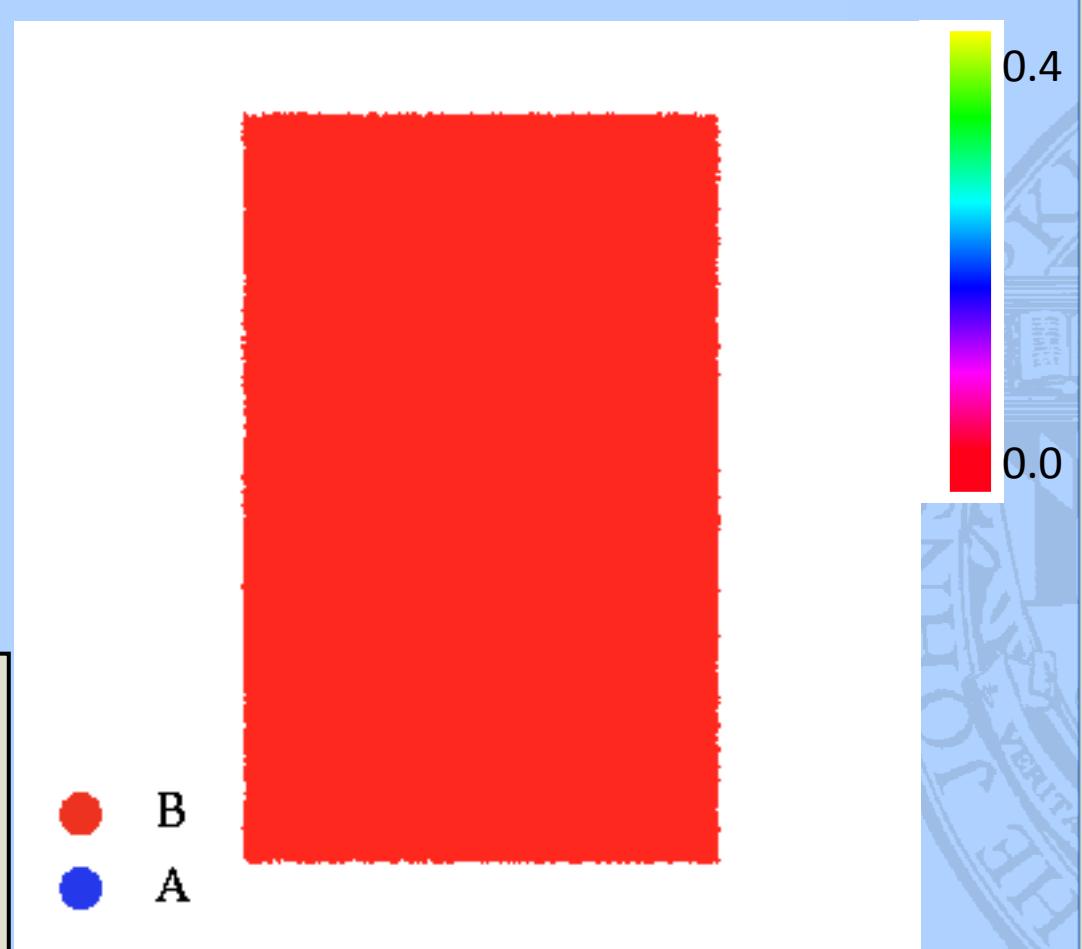
- 3D binary LJ 12-6 potential
- 80:20 composition, 144,000 atoms
- $T_{MCT} = 0.435$

Kob and Andersen, PRE 1995

# 3D Uniaxial Compression Test

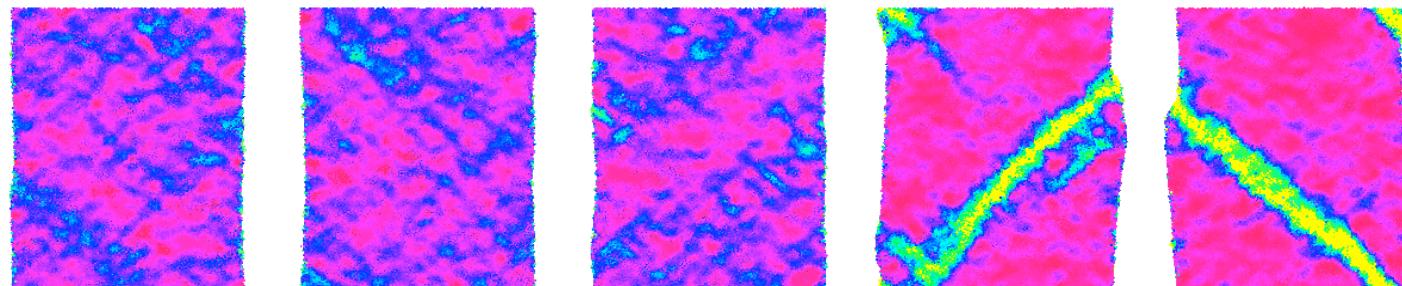


- Thin slab geometry to maximize in-plane spatial dimension  
 $75 \times 110 \times 15$ : 140,000 atoms
- Free surfaces in Y-Z
- PBC in X-Y and Y-Z
- Plane Strain: Average  $\sigma_{zz}$  zero



# Compression: Various Quenches

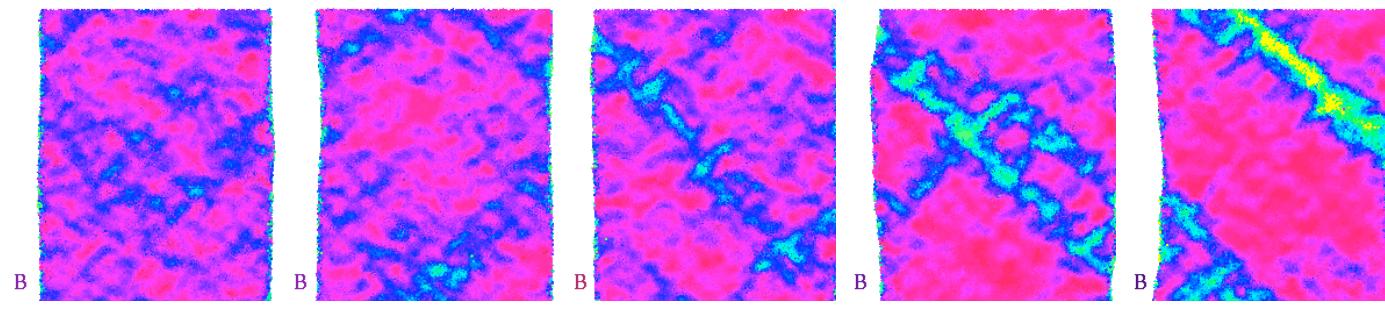
Dzugutov



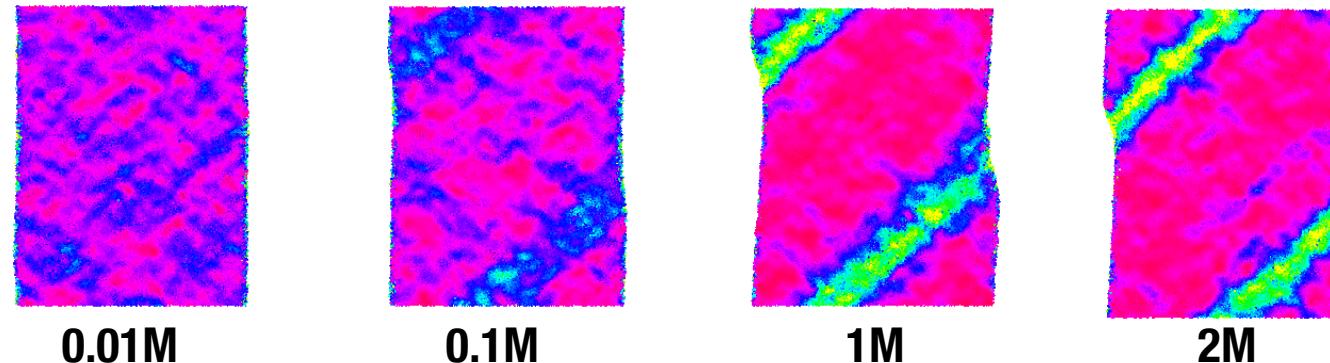
0.4

0.0

Wahnstrom LJ

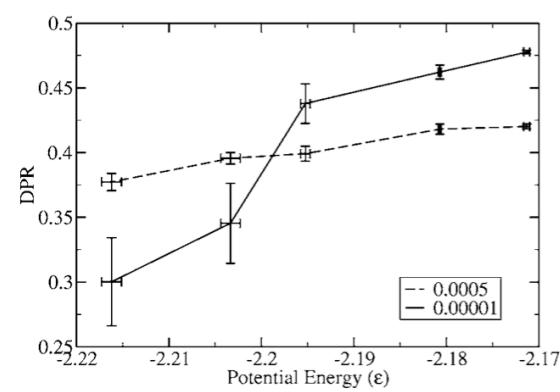


Kob-Andersen LJ

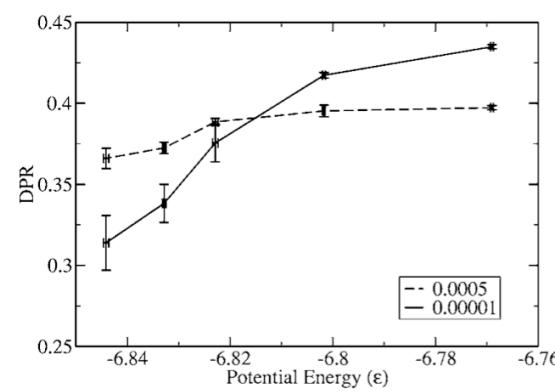


# DPR and Strain Rate Sensitivity

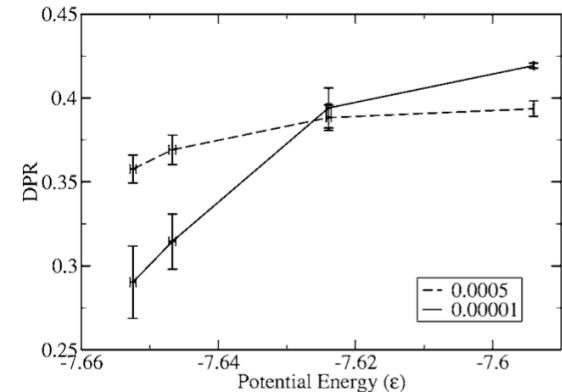
**Dzugutov System**



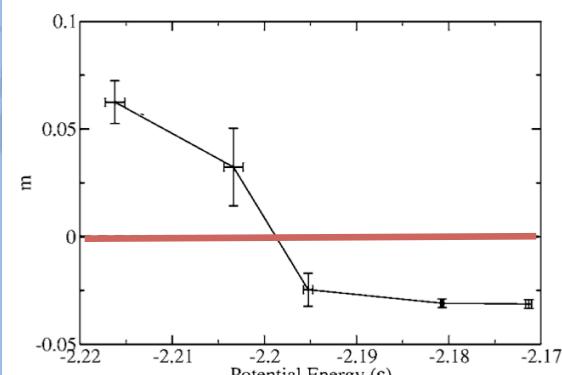
**Wahnstrom LJ**



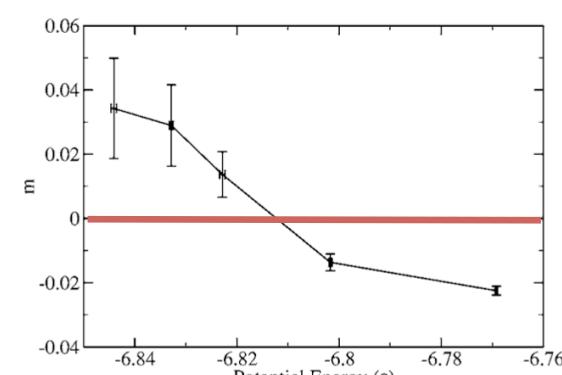
**Kob-Andersen LJ**



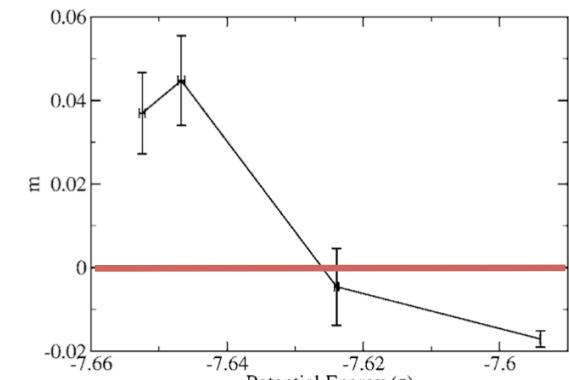
**PE**



**PE**



**PE**



# Triangulated Coordination Shell Analysis

Triangulated Coordination Shells: Bonds by atoms within the coordination shell form only triangles. The center atom and the triangle has to form a space dividing tetrahedral.

Criterion: (From Euler's formula)

$$\sum_q (6 - q)v_q = 12$$

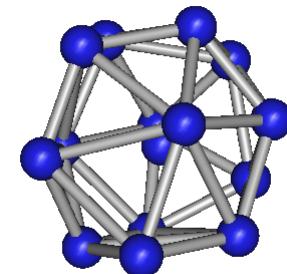
$v_q$  is the surface coordination number (from 3 to 8 for now)

$v_q$  is the count of neighbors has surface coordination number  $q$

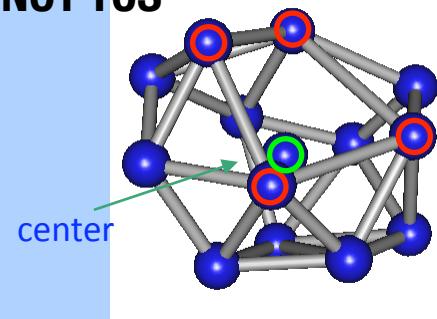
Glassy samples with lowest quenching rate

	TCS	Icosahedra
Dzugutov	25 %	12 %
Wahnstrom	13 %	10 %
K-A	3%	0.1 %

TCS

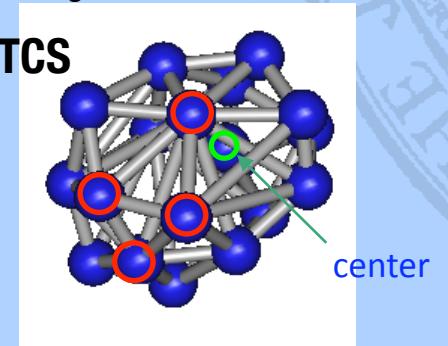


NOT TCS



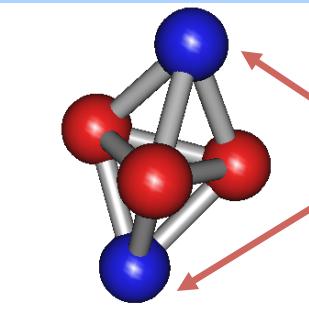
The four red atoms are forming a non-planar quadrilateral not a triangle

NOT TCS



The tetrahedral formed by 4 red atoms are space dividing but not consisting the center atom (green)

# 3D Percolation Analysis



**Connected!**

**Two atoms sharing at least three atoms are “connected”**

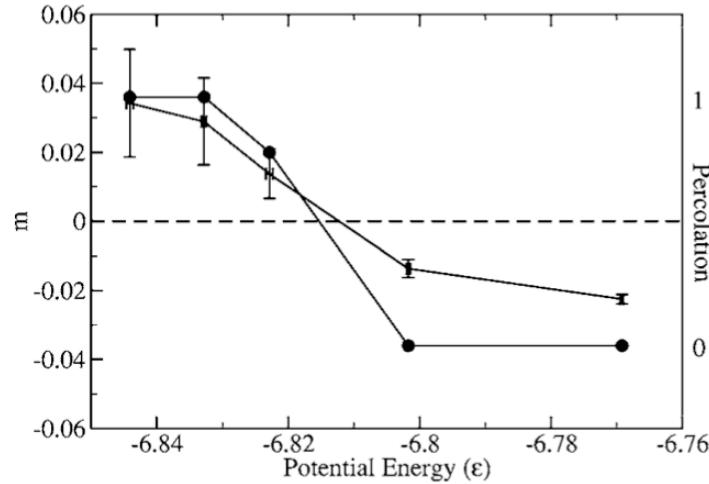
**The cage of two atoms have to interpenetrate or sharing faces**

**Similar to Zetterling, et al, JNCS, 2001**

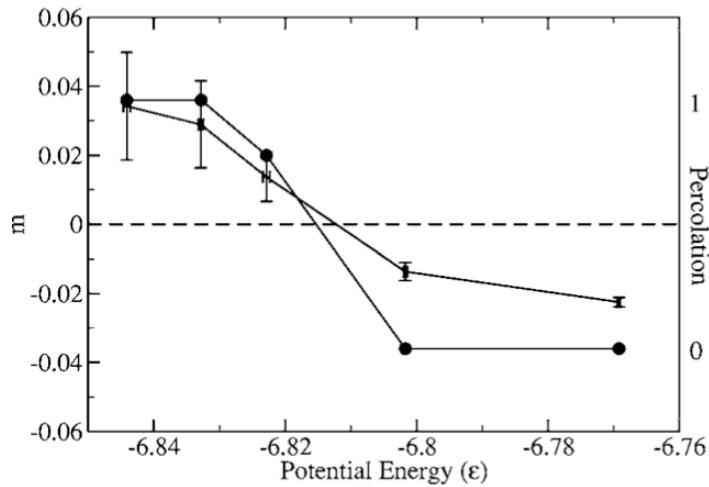
**DZ: (all percolate)**

**KA: (none percolate)**

**WA system (TCS SRO)**



**DZ system (Icosahedral SRO)**



# Observations

- **Experimental investigations of metallic glass reveal interesting behavior:**
  - Shear softening
  - Nonlinear Rheology
  - Systematic changes in ductility and creep dependent on alloy composition and sample history
  - Strong tendency to localization
- **Simulations confirm these and also reveal**
  - Plastic Hysteresis (Bauschinger effects)
  - Existence of a yield stress
  - Local nature of rearrangements
- **Recent investigations of shear banding show**
  - Slowly quenched samples more susceptible to localization
  - Samples that exhibit localization contain percolated short range order
  - CAVEAT: there is no universal method for defining short range order
- **Tomorrow: building a theory of plasticity**